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1 **Role of silica-based porous cellulose nanocrystals in improving water**
2 **absorption and mechanical properties**

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31

32 **Abstract:** Epoxy resins are important thermosetting polymers. They are widely used in
33 many applications i.e., adhesives, plastics, coatings, and sealers. Epoxy molding compounds
34 have attained dominance among common materials due to their excellent mechanical
35 properties. Here, the sol-gel simple method was applied to distinguish the impact on the
36 colloidal time. The properties were obtained with silica-based fillers to enable their
37 mechanical and thermal improvement. The work which we have done here on epoxy-based
38 nanocomposites was successfully modified. The purpose of this research was to look into the
39 effects of cellulose nanocrystals (CNCs) on various properties and applications. CNCs have
40 recently attracted a lot of interest in a variety of industries due to their high aspect ratio, and
41 low density which makes them perfect candidates. Adding different amounts of silica-based
42 nanocomposites to the epoxy system. Analyzed with different techniques such as Fourier-
43 transformed infrared spectroscopy (FTIR) and thermogravimetric analysis (TGA), scanning
44 electronic microscopic (SEM) to investigate the morphological properties of modified
45 composites. The various %-age of silica composite was prepared in the epoxy system. The 20
46 % of silica was shown greater enhancement and improvement. They show a better result than
47 D-400 epoxy. Increasing the silica, the transparency of the films decreased, because
48 clustering appears. This shows that the broad use of CNCs in environmental engineering
49 applications is possible, particularly for surface modification, which was evaluated for
50 qualities such as absorption and chemical resistant behavior.

51 **Keywords:** Bisphenol epoxy; D-400 epoxy; thermal properties; silica; polymer.

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57 **1. Introduction**

58 The most advanced technology, such as autos and electronics, are unable to match the unique
59 qualities of the materials. Metals, ceramics, polymers, and alloys are examples (Mubashir et
60 al., 2021). As a result of the enhancement of a number of features. It is necessary to create
61 two composite phases (Mubashir et al., 2018). The matrix and the reinforcement phase (Allie
62 et al., 2018; Aziz et al., 2020a; Jo et al., 2018). For the manufacturing of reinforcement
63 polymers, many types of organic or inorganic fillers are commonly used (Ali et al., 2020). By
64 this, the mechanical, thermal, electrical, and adhesive qualities have been enhanced (Ahmad
65 and Al-Sagheer, 2014; Aziz et al., 2019c; Park and Yun, 2018). In the epoxy system,
66 nanoparticles in a very small proportion are typically used for enhancement. Such
67 modification methods have an impact on inferior reduction processes including curing, heat
68 conduction, and mechanical supplies (Aziz et al., 2020c; Reig et al., 2018; Ullah et al.,
69 2021a). Several factors influence the characteristics of the micro polymer (Ali et al., 2021a).
70 Each component has fundamental qualities, such as shape, dimension, and the nature of its
71 interfaces (Aziz et al., 2019a; Messersmith and Giannelis, 1994; Salahuddin, 2004). Only a
72 modest amount of research on inorganic particles has been done in recent decades. The
73 ability to manufacture composites containing nanofillers, i.e., nanocomposites or
74 nanoparticles is the most significant (Aziz et al., 2020d; Zheng et al., 2021a). They're one of
75 the most hotly debated nano-dielectric systems right now. The fillers are the reinforcing
76 phases (Ali et al., 2021e; Aziz et al., 2019b; Tseng et al., 1999). Polymer-matrix composites
77 are employed in a considerable amount due to their ease of processing (Aziz et al., 2020b; Li
78 et al., 2018; Zhang et al., 2016). They are widely used in a variety of industries, including
79 electrical, automotive (Chuah et al., 2022a; Chuah et al., 2022b), and aerospace (Ahmad and
80 Guria, 2022). Polymer nanocomposites have received greater attention in recent years (Karim
81 et al., 2022). It serves as a replacement for predictable polymer composites (Aziz et al.,

82 2021c; Jamil et al., 2021; Li et al., 2016). The number of nano-size molecules in a polymer
83 composition can significantly improve its mechanical properties (Abbasi et al., 2023; Arshad
84 et al., 2023; Munir et al., 2023). This is mostly due to the nanoparticle's exclusive holdings,
85 which result in effective development properties (Kim et al., 2013; Longmire et al., 2008;
86 Ullah et al., 2021b). Polymer nanocomposites are materials in which at least one dimension
87 of the matrix is strengthened by particles (Aziz et al., 2021a; Aziz et al., 2021d; Chen et al.,
88 2019a). As the reinforced portion, conservative polymer composites are used in the
89 micrometer range (Al Sheheri et al., 2019; Ali et al., 2021d; Chen et al., 2019b). The bulk
90 polymer has a close relationship with the minor particle filling proportion and the composite
91 quality. As a result, the developed filler filling ratio is often required to achieve significant
92 improvements (Alston et al., 2019; Aziz et al., 2021b; Hassanzadeh-Aghdam et al., 2019).
93 Soft rubber particles and silica particles are examples of small collective particles. However,
94 polymer nanocomposites have received more attention than traditional polymers (Ali et al.,
95 2021c).

96 This remarkable enhancement began to be associated with the high surface range (Ahmad et
97 al., 2022). Between nanofillers and the polymer matrix, the volume proportion of
98 nanoparticles sets a substantial limit (Klonos et al., 2019; Li et al., 2018). Researchers have
99 also discovered that nanometer-scale fillers have greater active reinforcing characteristics
100 than micrometer-scale fillers (Ali et al., 2021b; Pan et al., 2019; Terzic et al., 2019). For these
101 active reinforcing effects, high definite surface area is largely responsible. As a result, the
102 area of interface with the polymer matrix affects efficient transmission of load from the
103 outside (Xie et al., 2018; Zotti et al., 2019a; Zotti et al., 2019b). Nanoparticles surface of
104 various materials makes them more valuable is one of the basic aspects of nanotechnology.
105 Geometry, filler kind, crystallinity, surface treatment, and degree of dispersion all influence
106 attributes of nanocomposites. The use of several types of fillers to agglomerate polymeric

107 materials improves their characteristics. The polymerization technique was used to modify
108 the surface of silica for epoxy molding compounds (EMC), such treatment was explored
109 comprehensively in this work (Ahmad et al., 2023). The current research focuses on epoxy-
110 based nanocomposites. The use of a Fourier transformed infrared spectroscope (FTIR) and
111 thermogravimetric measurements proved that the particles surfaces had been successfully
112 modified. To check and confirm the structural qualities of nanoparticles and nanocomposites,
113 several analysis and techniques were used.

114 Cellulose nanocrystals are nano-sized components that are biodegradable, biocompatible, and
115 renewable. It offers mechanical strength and stability. Environmental contamination is
116 reduced by employing various bioremediation techniques. The mechanism happening
117 between the contaminant and CNCs adsorbent should be developed in effective new
118 bioremediation strategies. CNCs structural functionalization helps to modify the
119 nanocellulose structure based on which it can be utilized for specific functions. Exploring the
120 mechanisms that contribute to the implementation of CNCs helps further developments and
121 advancement in biomedical applications.

122 Nanocellulose has drawn a lot of investigation over the last decade since. It is a plentiful
123 natural nanomaterial with a lot of potential. They are formed into multiphase order after
124 dispersing in a polar solvent, resulting in excellent optical and structural characteristics.
125 Significant progress has been made in addressing multi-dimensional qualities and a wide
126 range of high-tech applications of cellulose nanocrystals (CNCs) in many domains, according
127 to the report. In the long run, these qualities have an impact on the composite's overall
128 performance. Cellulose is a polysaccharide macromolecule with a linear structure. Through
129 Vander wall contacts and hydrogen bonds, these macromolecules combine and form
130 microfibrils that are aligned and stabilized horizontally, resulting in crystal and non-
131 crystalline areas. As a result, CNCs in all walks of life play a vital role in numerous industrial

132 applications to provide a complicated approach in the future, as shown in this preliminary
133 study. We present native cellulose nanocrystals that have undergone surface modification
134 through the use of a silane coupling agent as shown in Figure 1.

135

136 **2. Experimental work**

137 **2.1. Materials and methods**

138 Bisphenol epoxy or E-51 (DGEBA) commercially available epoxy.

139 Diamine D-400 also commercially available epoxy,

140 3-aminophenyltetraethylorthosilicate (APTES) were supplied from (Thermo scientific)

141 employed as a cross-linker were used in this experiment.

142 The Aladdin Shanghai industry in China provided the highly pure analytical ethanol absolute
143 (purity 99.7%).

144 Petri dishes, hot air oven, distilled water,

145 Silica powder from DINGLONG QUARTZ, China, purity of 99.3%.

146 During the entire experiment, distilled water was used, as well as other lab equipment. The E-
147 51 epoxy was made by reacting an equimolar quantity with diamine at a 4:2 ratio. In the
148 round bottom flask, 4g of E-51 epoxy and 2g of diamine were combined for this reason. The
149 mixture was then swirled for 3 hours at normal temperature. After forming a clear and
150 transparent solution, it was poured into 2-3 clean containers Petri dishes. These containers
151 were then carefully placed and baked for 120 minutes at 100°C in a hot air oven. After
152 curing, another 45 minutes at 60 °C were put in vacuum oven. A flexible thin film with a
153 consistent of 1cm was cast. FTIR, TGA and Raman techniques were characterized on the
154 same day. Two different types of composite systems were created, with a different quantity of
155 silica. Each system was given a distinctive abbreviation. The measured phase separation and
156 the percentage of silica are given in Table 1.

157 **2.2. Epoxy-silica composite material and casting thin films**

158 Silica composite films were obtained with varying %-ages using hydrolysis methods. The
159 separation of silica was found to be 20 percent. As the silica content increased the
160 transparency of the films reduced. Epoxy-silica composite film was found with consistent
161 thickness. Epoxy-silica systems were created by pouring them into petri dishes with a flat
162 surface.

163

164 **2.3. Epoxy-silica composite coupled and uncoupled synthesis**

165 In a separate beaker, the uncoupled epoxy-silica composites were carefully weighed. The E-
166 51 epoxies and tetramethylorthosilicate (TMOS) were added. Stirred for an hour at normal
167 room temperature with an anhydrous state. After obtaining a clear mixture, a stoichiometric
168 amount of water was added and swirled at room temperature for another 2 hours. The
169 temperature was increased up to 80 °C and swirled it for another 8 hours. After that, diamines
170 were added to the epoxy system as a curing agent and mixed for 1 hour. The mixture was
171 then poured and baked at 100°C for 90 minutes into petri dishes for curing process
172 completion. The flexible films were obtained with a uniform thickness of 1cm (Ali et al.,
173 2021b; Zheng et al., 2021b). APTES was combined with TMOS to chemically connect the
174 inorganic phase. On one side, the APTES combined with silica, while on the other, it reacted
175 with the secondary amino group of the epoxy. As a result, chemical linkages were formed
176 between two phases in APTES (Ahmed et al., 2019; Ahmed et al., 2021). This improved
177 system, which is presented in Table 2 was used to make composites with varied percentages
178 of silica, just like an uncoupled system. E-51 epoxy system, 3-APTES, and epoxy-silica
179 linked was also carefully weighed. This mixture was added and stirred for one hour
180 continually to form a solution in glass plates. Tetraethylorthosilicate and distilled water was
181 added drop by drop and stirred the mixture for the next 5 hours continually at temp; 60 °C to

182 maintained the solution. Diamine was added to the mixture at a calculated amount, which was
183 then agitated for 60 mins until it became homogeneous. To eliminate the contaminants, the
184 mixture was poured onto Petri plates and baked at 100 degrees Celsius. At 110°C, films were
185 then cured for 1 hour. A constant thickness of 1cm was used to make the solid films.
186 Different techniques, including as FTIR, Raman, and TGA, were used to characterize the
187 films.

188

189 **2.4. Characterization of the material**

190 Various analytical approaches were used to characterize epoxy-based silica nanocomposites
191 systems with varying %-ages. The morphology of polymer nanocomposites with and without
192 the influence of APTES was investigated. It provides qualitative information regarding
193 conversion of monomer-to-polymer. Fourier transform infrared spectroscopy (FTIR) Nicolet-
194 5700 spectrophotometer by the KBr-pellet method and scanned from 4000-500 cm^{-1} with a
195 resolution superior to 0.5 cm^{-1} . Exciting light of 532 or 633 nm is used to perform. Raman
196 spectroscopy measurements are performed at room temperature, using exciting radiation of
197 532 or 633 nm, with an incident power included between 0.04 and 5 mW. Raman spectrum of
198 polymer nanocomposites is characterized by three typical bands, namely at 1350 cm^{-1} the D
199 band, at 1580 cm^{-1} the G band, and the 2D band at a frequency at 2700 cm^{-1} . The D band is
200 related to the breathing mode of carbon atoms on the aromatic ring and is not always visible;
201 the G peak is determined by in-plane optical vibrations of the sp^2 -bonded carbon atoms,
202 whereas the 2D peak is approximately double the frequency of the D band as well as arises
203 from second-order Raman dispersion process. Thermal gravimetric analyses (TGA) TA,
204 Q200 were used. For stirring Magnetic stirrer 85-1 type Zhi Wei Shanghai was used. (Cence
205 TG16-WS) Hunan Xiangyi Laboratory Instrumental Development Co. Ltd was used as
206 centrifuged for centrifugation. The thermal stability of the polymer films in these

207 nanocomposites added the creation of an inorganic network according to the findings. As the
208 amount of silica in the system increased. The deterioration process has slowed as the TMOS
209 content has increased.

210

211 **3. Results and discussion**

212 **3.1. FTIR spectra of cured systems**

213 FTIR spectrum nanocomposites system in Figure 2, reveals a variety of peaks, revealing vital
214 information about SiO₂-epoxy. The removal of specific epoxide and amino functional group
215 confirmed the creation peaks of the epoxy network. O-H (str) at 3741 cm⁻¹ and N-H (str) at
216 3642 cm⁻¹ functional groups, respectively (Zhang et al., 2021).

217 At 2968 cm⁻¹ a peak develops while C-H with extending vibration. The C-H stretch at 925
218 cm⁻¹, confirms the successful formation. In the functional group regions, cured
219 nanocomposites systems show some comparable peaks. The asymmetric stretching vibration
220 peak of C-H is located at 2966 cm⁻¹, as can be seen in Figure 3. The band appeared at 1452
221 cm⁻¹ corresponding to CH₂ in epoxide moiety. The band generated at 1775 cm⁻¹ attributed to
222 the carbonyl group (C=O) in epoxide network. Similarly, the band for Si-O-Si symmetric
223 vibrations was appeared at 129 cm⁻¹.

224

225 **3.2. Raman spectra of nanocomposites system**

226 To examine the chemical groups in the nanocomposites system, Raman spectra were also
227 study. Several Raman spectra bands conforming the epoxide vibration in the 1230cm⁻¹ to
228 1280 cm⁻¹ region. At 1250 cm⁻¹. The epoxide groups absorption in the epoxy resin mixture
229 determines the intensity of this peak. The epoxide ring deformation appears to be weaker at
230 912 cm⁻¹. At 1118 cm⁻¹, the other Raman peaks show epoxy resin backing vibrations.

231 Throughout the curing process, they maintain the same intensity. The oxirane ring was
232 designated at 3070 cm^{-1} of C-H stretching peak are shown in Figure 4.

233

234 **3.3 Thermal stability of composites and nanocomposites system**

235 The polymer composite shows thermal deterioration with a maximum temperature (T_{\max}) of
236 $539\text{ }^{\circ}\text{C}$. There was no evidence of silica content. The thermogravimetric analysis curve
237 begins at $220\text{ }^{\circ}\text{C}$ and gradually decreases in weight until it reaches $427\text{ }^{\circ}\text{C}$ due to TMOS
238 moisture condensation reaction absorption. Because open-chain carbon has a lower energy of
239 degradation than closed-chain carbon (Chuah et al., 2022c; Hii et al., 2009; Cheah et al.,
240 2016; Chuah et al., 2021). Figure 5 shows the curve of weight loss abrupt fall from $500\text{ }^{\circ}\text{C}$ to
241 $519\text{ }^{\circ}\text{C}$, indicating that open chain carbon combustion begins about $450\text{ }^{\circ}\text{C}$. For silica-based
242 epoxy, a two-step degrading mechanism was investigated. The neat epoxy composite
243 polymer degradation temperature was raised. With the enormous weight loss compared to
244 silica loadings, the thermal degradation temp; increased at 15% of silica. This research
245 increased the thermal stability and organization of nanocomposites at high temperatures. All
246 uncoupled silica-epoxy systems experience a reduction due to the presence of a silica
247 network. Thermal breakdown of macromolecular chains is further slowed by the reaction of
248 the thermograms of the uncoupled nanocomposites system as shown in Figure 6.

249

250 **3.4 CNCs modification**

251 Round bottom flask of 150 mL with 100 mL distilled water and 5 mL (KH-560) were added.
252 For adjust the pH to 4, acetic acid was added to the mixture and stirred for 60 mints. 0.3 g of
253 CNCs were added to the mixture and sonicated for thirty minutes were used for dispersion.
254 The mixture was then agitated for 150 minutes at room temp; To remove excess silane, it was
255 washed twice by centrifugation techniques with ethanol and once with distilled water. It was

256 then dried for 6 hours at 80°C in an oven. The crystals were identified as MCNCs and
257 preserved in a glass vial for further use.

258 One of the simple ways to obtain materials with the appropriate qualities is to purchase
259 surface changes from various natural and synthetic resources through various treatments
260 (Dhakal et al., 2012). The effects of monomers on particle size have been studied in the past
261 (Jiménez Saelices et al., 2019). However, research on the effects of KH-560 treatment on the
262 physical and chemical properties of CNCs is limited (Javanbakht et al., 2016a; Javanbakht et
263 al., 2016b). As a result, the effects of KH-560 treatment on physical and chemical
264 parameters, such as CNC water absorption, were investigated in this study. The CNCs' OH
265 group was discovered to be responsible for water/moisture absorption. These functional
266 groups, which consist of a linear polymer lacking water units joined by the glycosides bond,
267 playing a crucial role in the surface modification of CNCs (Sun et al., 2005; Thakur et al.,
268 2013). Mercerization is the most effective method for removing contaminants from natural
269 cellulose's surface. The CNCs function group is started utilizing the standard manner
270 described in the literature in this study. After that, the CNCs are changed with a silane
271 coupling agent.

272

273 **3.5 CNCs with coupling agent**

274 Figure 7 depicts the KH-560 modified with CNCs. OH, groups are present on the surface
275 prior to modification. The peak of the Si-O-C bond appeared at 1028 cm⁻¹. The existence of
276 the silane functional groups was clearly demonstrated by the peaks at 1575 cm⁻¹. It signifies
277 that the surface of CNCs was successfully changed with KH-560 silane (Abraham et al.,
278 2016; Xu et al., 2013).

279

280 **3.6 Thermal properties**

281 The cellulose nanocrystals (native and modified) are a dynamic approach to investigating the
282 response to temperature change. Native cellulose nanocrystals have a different thermal
283 behavior than modified cellulose nanocrystals (Li et al., 2011; Ma et al., 2017). Figure 8
284 demonstrates that thermal deterioration occurs at a higher temp; range in CNCs, owing to its
285 size, excellent thermal stability, and a considerable drop in molecular weight degradation in
286 the high amorphous regions. The disintegration of the CNCs follows a usual pattern, with
287 temperatures ranging from little above 340-460°C, with a minor quantity of ash remaining at
288 600°C. The insertion of sulfate groups by hydrolysis found considerably lower the activation
289 energy of CNCs breakdown (Feng et al., 2017; Lu and Hsieh, 2010). Furthermore, the
290 disintegration of CNCs at low temp; 370°C implies have a quicker heat transmission rate.
291 The thermal conductivity of the (NCNCs and MCNCs) mixtures is nearly identical, while the
292 modified mixture's thermal conductivity is lower than that of the native ones (Fortunati et al.,
293 2012; Kumar et al., 2014).

294

295 **3.7 Scanning electron microscopy**

296 CNCs particularly unaltered are brittle and easily shattered. The SEM image shows a broken
297 cross-section surface of the CNCs. The surface layers are apparent in which the orientation of
298 the nanocrystals is clearly seen exhibit reasonable homogeneous mixture as shown in Figure
299 9A. Agglomeration happens when more cellulose nanocrystals are added (MCNCs) as
300 illustrated in Figure 9B. Dispersion of modified cellulose nanocrystals also appear on the
301 surface. A micro-sized interfacial interaction between MCNCs and epoxy system could be
302 visible (Espino-Perez et al., 2016; Majoinen et al., 2012; Peng et al., 2017). It indicates that
303 the rough surface layer on the surface of modified CNCs materials treated with KH-560

304 exhibits a significant improvement. The elimination of contaminants is responsible for
305 morphological alterations.

306

307 **3.8 Water absorption study**

308 The water absorption of epoxy materials has been a research topic in many studies (Chow,
309 2007; Takeshita et al., 2014). The improved performance of epoxy materials can be
310 considerably influenced by the amount of water absorbed; its structural applications are
311 limited. Combining natural hydrophilic elements as enhancers, such as CNCs, may
312 exacerbate moisture absorption and have a negative impact on nanocomposites'
313 characteristics. Grafting boosted the water uptake capacity, according to reports (Cankaya
314 and Temuz, 2015; Kumar et al., 2019; Mekonnen et al., 2013). CNCs hydrophobic
315 transformation has the ability to alleviate this fundamental issue. This research looked at how
316 adding native and enhanced CNCs to epoxy composites influences water ingress. Within
317 seven days, the water absorption mode of native and modified CNCs is incorporated into
318 clean epoxy resin and nanocomposite materials. Different CNCs are processed with the KH-
319 560 in accordance with the approach described in the experimental section as shown in
320 Figure 10. CNCs surface functionalization using the KH-560 is a simple approach to modify
321 the surface properties. Because hydrophobic polymers are grafted to CNCs, it is not as
322 successful as transplant copolymer technique. Hydrolysis, condensation, and bond formation
323 are all functions of the KH-560 CNCs. One of the most practical ways to produce materials
324 with the desired qualities is to modify the surface of materials purchased from various natural
325 and manmade resources using various treatments. The effects of KH-560 treatment on the
326 physical and chemical properties of CNCs have been examined, as well as the impacts of
327 KH-560 treatment on the mechanical properties of polymer composites. As a result, here
328 investigated the impacts of KH-560 treatment on physical and chemical parameters, such as

329 CNCs water intake. Moisture absorption was discovered to be caused by hydroxyl groups
330 contained in the backbone of CNCs. Because such substrates are made up of linear with
331 repeating glucopyranose units. These functional groups are also important in surface
332 modification. The function group of CNCs is launched in this work utilizing the standard
333 approach and the reinforcement method of various concentrations. After that, silane couplers
334 are used to surface-functionalize the modified CNCs.

335 On CNCs, the KH-560 coupler was found to be the most efficient. Chemical resistance is
336 another feature of these functional CNCs. The functionalization of silane improves the
337 chemical resistance of CNCs. On CNCs that are vulnerable to chemical attacks, this behavior
338 is likewise attributed to the blockage of the active function group. Water is dependent on the
339 existence of the polymer matrix's midstream. (Becker et al., 2004; Jahan et al., 2018). By
340 taking up available space, nanoparticles can usually restrict their acceptance. The kind,
341 concentration, and function of nanoparticles, on the other hand, are critical in preventing
342 moisture absorption. The absorption was higher when primary CNCs and modified ones were
343 used instead of the neat epoxy matrix concentration range. Higher grafted modified CNCs,
344 considerably enhanced water absorption relative to unmodified CNCs, while unmodified
345 CNCs' moisture absorption reduced. Nanocomposites have a much lower moisture absorption
346 rate than native CNCs, and their load is just 1% of that of modified CNCs. It becomes
347 hydrophobic when the OH function is replaced by hydrophobicity. It appears that the
348 modified CNCs' hydrophobic qualities are transmitted to epoxy nanocomposites, resulting in
349 a reduction in water absorption.

350

351 **4. Conclusion**

352 Hydrolysis is a critical method. Different silica-based epoxy nanocomposites were used in
353 this study. The CNCs function group is started utilizing the standard manner described in the

354 literature in this study. After that, the CNCs are changed with a silane coupling agent. The
355 existence of the silane functional groups was clearly demonstrated by the peaks at 1575 cm^{-1} .
356 The thermal characteristics of the modified nanocomposites have been significantly
357 improved. The surface layers are apparent in which the orientation of the nanocrystals is
358 clearly seen exhibit reasonable homogeneous mixture. CNCs, with KH-560 coupler was
359 found to be the most efficient. The functionalization of silane improves the chemical
360 resistance of CNCs. The epoxy properties are developed by using an APTES coupling agent
361 with silica. The interfacial contact was improved by using silica-based nanocomposites. The
362 20 % of silica was shown greater enhancement and improvement. They show a better result
363 than D-400 epoxy. The temperature has a significant effect on changed silica over the course
364 of the experiment. The rate of hydrolysis rises with time, as does the alteration of silica.

365

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375

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377

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