



Peng, Y. M., Unluer, C. and Shi, J.Y. (2021) Rheo-viscoelastic behavior and viscosity prediction of calcium sulphoaluminate modified Portland cement pastes. *Powder Technology*, 391, pp. 344-352.
(doi: [10.1016/j.powtec.2021.06.023](https://doi.org/10.1016/j.powtec.2021.06.023))

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Deposited on 28 June 2021

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1 Rheo-viscoelastic behavior and viscosity prediction of calcium
2 sulphoaluminate modified Portland cement pastes

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7

8 **Highlights:**

- 9 • A model to predict the η_{\min} by PC and CSA volume fractions is proposed.
- 10 • PC and CSA increase the η_{\min} , critical strain and flow strain of paste.
- 11 • The proposed η_{\min} prediction model has high accuracy at both 5 min and 60 min.
- 12 • The accuracy of prediction model at 5 min is higher than that at 60 min.

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14

15 **Abstract:** In this study, the rheo-viscoelastic behaviors of calcium sulphoaluminate (CSA) modified
16 Portland cement (PC) pastes were investigated. The shear stress, apparent viscosity, frequency
17 sweep and time sweep curves of composite pastes with different PC and CSA volume fractions were
18 analyzed, and the minimum apparent viscosity (η_{\min}) of PC-CSA pastes were predicted. Results
19 show that with the increase of PC and CSA volume fractions, η_{\min} , critical strain and flow strain of
20 pastes increase at both 5 min and 60 min resting time, and the phase angle varies under different
21 time spots. The accuracy of viscosity prediction models for 5 min and 60 min are both relatively
22 high. Meanwhile, the reason why the prediction accuracy of 5 min is higher than 60 min is that with

23 the prolongation of resting time, the relationships between particle volume fractions and rheological
24 properties are affected by the early microstructure formed by flocs and hydration products, thus
25 affecting their original functional relationship at 5 min.

26

27 **Keywords:** Portland cement, calcium sulphoaluminate cement, minimum apparent viscosity,
28 viscoelasticity, viscosity prediction model

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30

31 **1 Introduction**

32 With the development of cement industry, it has gradually become an effective method to improve
33 specific properties of cement mixes by mixing with other kinds of mineral admixtures or clinkers.
34 Among them, calcium sulphoaluminate (CSA) modified Portland cement (PC) is considered as a
35 kind of compound cement with great potential and prospect due to its advantages of combination of
36 both PC and CSA clinkers [1].

37 Owing to the high production of carbon dioxide and large consumption of energy in the process
38 of cement clinker calcination, reducing the utilization of cement clinker has important practical
39 significance for energy saving and environmental protection. CSA clinker can be made of limestone,
40 calcium sulfate and alumina as well as some industrial by-products or wastes, leading to its lower
41 CO₂ emission per gram of clinker than that of alite, which is a main component of PC clinker [2-4].
42 The maximum calcination temperature of CSA is 200°C lower than PC clinker, and CSA clinker is
43 easy to grind compared with PC [5, 6]. Furthermore, CSA cement also has the characteristics of
44 high early strength development rate and shrinkage compensation [7]. From these points of view,

45 PC-CSA mixes has great development potential. Many researchers have carried out relevant
46 researches on fresh and hardened properties of PC-CSA mixes [8-9]. After incorporating CSA, the
47 setting time of cement paste decreases [8]. CSA clinker plays an important role in early mechanical
48 properties, while OPC is mainly responsible for hardened strength [9]. Meanwhile, curing
49 temperature is critical to the hydration process of CSA cement. Different curing temperature will
50 not change the type of hydration products of CSA cement, but will affect its quantity [10]. In terms
51 of raw materials for the production of CSA clinker, Shen et al. [11] studied the potential of using
52 phosphogypsum (PG) to produce CSA cement and the decomposition behavior of PG in CSA
53 cement production, and the results showed that PG can be used to produce CSA cement, and the
54 effect of PG decomposition on the formation of CSA is significant. In addition, the incorporation of
55 mineral admixtures and chemical admixtures can also affect CSA cement or PC-CSA cement. The
56 early strength and setting time of CSA can be significantly enhanced by lithium carbonate-aluminum
57 sulfate at 0 °C [12]. At the same time, lithium carbonate aluminum-sulfate not only promoted the
58 formation of ettringite, but also accelerated the hydration of C₂S. Zhang et al. [13] found that the
59 addition of excessive retarder will reduce the activity of CSA cement particles and contribute to the
60 decline of compressive strength. Fu et al. [14] pointed out that burnt plaster and alunite can control
61 the decrease of long-term strength of sulphoaluminate cement, and they can also improve its
62 resistance to drying contract and chemical correlation.

63 Rheological properties of fresh cementitious materials not only have an important impact on the
64 construction process, but also are closely related to the hardened performances [15]. Rheological
65 properties are the intrinsic properties of concrete workability, and rheology of cement paste largely
66 dictates rheology of concrete, especially at early ages [16, 17]. For the Bingham model, the two

67 most important basic parameters are plastic viscosity and yield stress [18, 19]. Besides the properties
68 of fluid, cement paste will show certain elastic properties when subjected to different amplitude or
69 frequency of external shear force, which is described as the term “viscoelasticity” [20, 21]. With
70 the continuous hydration of cement, the paste gradually loses its fluidity, sets and hardens into a
71 semi-solid or solid-phase system dominated by elasticity [22, 23]. Rheological properties and
72 viscoelasticity (rheo-viscoelastic properties) summarize the change process of the physical state of
73 paste from fresh stage to setting and hardening stage. The addition of several additives can also
74 affect the rheological properties of PC or CSA paste. Hydroxypropyl methylcellulose could
75 significantly increase the stress and viscosity of sulphoaluminate cement paste [24]. The hydration
76 rate and compressive strength of CSA mixes were improved after the incorporation of ettringite seed
77 crystals, while yield stress, plastic viscosity and setting time was on the decline [25].

78 CSA modified PC pastes have excellent properties of both PC and CSA. Despite many
79 researches have focused on the effects of mineral or chemical admixtures on rheology of PC or CSA
80 mixes, there are still few comprehensive studies on the rheo-viscoelastic of PC-CSA composite
81 pastes. Considering that there is also no relevant viscosity prediction analysis for this system, in this
82 study the rheo-viscoelastic behavior is comprehensively analyzed by means of dynamic rheological
83 test, amplitude sweep and time sweep, and then the prediction model of the minimum apparent
84 viscosity (η_{\min}) under the resting time of 5 min and 60 min was given. Finally, based on the
85 viscoelasticity test within 60 min, the reasons for the difference in the prediction accuracy of
86 viscosity under different resting time were discussed.

87 **2 Materials and methods**

88 **2.1 Materials and mixture proportions**

89 Commercial Portland cement (PC) P-I 42.5 (conforming to GB 8076) and calcium
90 sulphoaluminate cement (CSA) were provided by China Building Materials Academy. The density
91 of PC and CSA is 3.15 g/cm³ and 2.80 g/cm³ respectively. Their chemical composition and some
92 physical properties are shown in **Table 1**, and particle size distribution is presented in **Fig. 1**.
93 Polycarboxylate superplasticizer (PCE) with water-reducing rate 32% and solid content 33.1% was
94 used to disperse fresh composite pastes.

95 The mixture proportions of PC-CSA systems are displayed in **Table 2**. In order to improve the
96 accuracy of prediction, here the mixed PC and water are regarded as a solvent, while CSA is
97 considered as a solute to predict the viscosity of PC-CSA-H₂O paste (**Fig. 2**). According to the mass
98 ratio of **Table 2**, when the volume ratio $\varphi_C/(\varphi_C+\varphi_W)$ (abbreviated as $\varphi_{C/W}$) is between 0.45 and 0.6,
99 the water-to-binder ratios (W/B) of composite pastes incorporating less than 20wt % CSA range
100 from 0.15 to 0.4. Because some of mix proportions have very low W/B ratio and a certain volume
101 of flocculation structure exists in the fresh pastes, which may affect the relationship between particle
102 volume fraction and viscosity, a sufficient amount of PCE was added into each paste to disperse the
103 flocs and ensure the homogeneity of paste. The process of mixing pastes was first mixing at 60 rpm
104 for 60 s, then stopped for 30 s, and finally mixing at 120 rpm for 60 s.

105 **2.2 Rheological test**

106 **2.2.1 Dynamic test**

107 An Anton Paar MCR 102 rheometer was used to measure the rheology of different pastes. The
108 height (H) of 4-blade vane rotor was 40 mm, and the internal radius (R_1) and outer radius (R_2) were

109 11 mm and 21 mm respectively. The external cylinder remained stationary during rheological test.

110 The rheological test protocol is shown in **Fig. 3**. The test started at 5 min after the initial contact
111 of water, PC and CSA, and the test was conducted again after 60 min. First, a 60 s pre-shear at 100
112 s^{-1} were intended to create a uniform condition. After resting for 60 s, the shear rate increased
113 logarithmically from 1 s^{-1} to 100 s^{-1} in 100 s, and the time interval between two test points was 2 s.
114 The typical rheological curves of pastes are shown in **Fig. 4**. It is found that with the increase of
115 shear rate, the rheological behavior will change from shear thinning to shear thickening. This shear
116 rate is called critical shear rate ($\dot{\gamma}_{crit}$), and the corresponding shear stress and viscosity can be
117 described as critical shear stress (τ_{crit}) and η_{min} [26]. When the apparent viscosity of paste reaches
118 the minimum, the volume of flocculation structure is relatively small and the relationship between
119 the volume fraction and rheological properties is significant. Therefore, the aim of this study is to
120 predict the η_{min} of composite pastes with particle volume fractions of PC and CSA.

121 2.2.2 Amplitude sweep test

122 The goal of amplitude sweep is to measure the change of viscoelasticity with strain. In oscillation
123 test, the rotor oscillates along the axis at a set frequency. Here the frequency was kept constant at 1
124 Hz, while the strain was increased logarithmically from 10^{-3} % to 10^2 % in 0 s~200 s (**Fig. 5**). The
125 test was carried out at 10 min after the contact of clinker and water. The variation of some
126 viscoelastic parameters with amplitude can be obtained by amplitude sweep test, such as G' , G'' and
127 δ . The relationships among these parameters are as follows [20]:

$$128 \quad G' = \frac{\sigma_0 \cos \delta}{\varepsilon_0} \quad (1)$$

$$129 \quad G'' = \frac{\sigma_0 \sin \delta}{\varepsilon_0} \quad (2)$$

130
$$\tan \delta = \frac{G''}{G'} \quad (3)$$

131 Where G' is storage modulus, G'' is loss/viscous modulus, σ_0 is stress, δ is phase angle and ε_0 is
132 strain. In these parameters, G' characterizes the elastic behavior of paste, while G'' characterizes the
133 viscous behavior.

134 **2.2.3 Time sweep test**

135 The change of viscoelasticity with time due to the evolution of paste internal microstructure can
136 be investigated by time sweep test. In this study, the same frequency (1 Hz) as the amplitude sweep
137 was adopted, and the results of amplitude sweep showed that for PC-CSA system, when the
138 amplitude was controlled within 10^{-4} %, both G' and G'' were relatively stable, and the internal
139 structure of fresh paste had not been significantly damaged, so the amplitude value was determined
140 to be 10^{-4} %. G'' and G' were measured every 30 s for 60 min, and the phase angle δ was calculated
141 by the tangent of G''/G' .

142 **3 Results and discussion**

143 **3.1 Rheo-viscoelastic behavior**

144 **3.1.1 minimum apparent viscosity**

145 **Fig. 6** give the change of η_{\min} with different PC and CSA particles volume fraction (i.e. $\varphi_{C/W}$
146 and φ_S). The change in η_{\min} were tested over 60 min after the initial contact of binders with water.
147 The changing trend reveals that η_{\min} depends not only on $\varphi_{C/W}$ and φ_S , but also relates to resting time.
148 As seen from **Fig. 6**, the incorporation of CSA dramatically increases the η_{\min} at both 5 min and 60
149 min resting time. The viscosity of composite pastes increases with increasing φ_S ranging from 0 to
150 0.2. Furthermore, with the increase of $\varphi_{C/W}$, η_{\min} of composite pastes also increases gradually. The
151 η_{\min} of PC-CSA pastes with $\varphi_{C/W}=0.45$ and $\varphi_{C/W}=0.5$ are very close. However, when $\varphi_{C/W}$ rises to

152 0.55, the viscosity increases significantly. When the particle volume fraction reaches a certain
153 degree, the rate of contact between particles and the formation of microstructure such as flocculation
154 and initial hydration products are accelerated, which may significantly increase the yield stress and
155 viscosity of cement pastes [27]. Compared with 5 min, the η_{\min} of composite pastes under the same
156 $\varphi_{C/W}$ and φ_S is larger at 60 min resting time.

157 3.1.2 Amplitude sweep

158 Viscoelasticity is an important property to characterize the change of internal structure of
159 pastes. **Fig. 7** presents the typical curves of G' , G'' and their evolutions with strain, and the
160 viscoelastic parameters of pastes with different PC and CSA volume fraction are summarized in
161 **Table 3**. Generally, if the test sample has a yield point, the G' which represents the elastic
162 behavior plays a dominant role in small deformation. In a certain deformation range, the two
163 moduli can be kept at a constant level, which is independent of the set amplitude and frequency.
164 This range of reversible elastic deformation is also known as the linear viscoelastic region (LVR)
165 [28]. Then G' and G'' are no longer constant and begins to decline, which correspond to the critical
166 strain. When G' and G'' curves intersect, the amplitude at the intersection of storage modulus and
167 loss modulus can be regarded as the flow point. When the applied strain is larger than this critical
168 value, the microstructure of paste will be destroyed irrecoverably. According to the variation of
169 viscoelastic parameter, it can be found that when $\varphi_{C/W}$ and φ_S increases, the values of G' and G''
170 corresponding to the critical amplitude and the critical strain/flow point strain all increase. The
171 wider the range of LVR is, the larger amplitude is needed to destroy the network structure of
172 colloidal particles in pastes, showing its stronger ability to resist external deformation. The
173 addition of CSA densifies the internal structure and make the pastes difficult to be disassembled.

174 3.1.3 Time sweep

175 **Fig. 8** illustrates the changing trend of G' , G'' and δ over time. Regardless the volume fraction
176 of PC and CSA, all pastes present similar results, that is, G' and G'' increased with increasing resting
177 time. For C45-S0 paste, the value of G' is greater than G'' within the first 50 min. δ fluctuates in the
178 first 30 min, and then increases with the prolongation of resting time after 30 min. When the volume
179 fraction of CSA increase (**Fig. 8(b)**), G' and G'' are higher than that of C45-S0 paste at the same
180 time point, and the two moduli are not intersected within 60 min. The decreasing rate of δ is more
181 obvious in the first 20 min, and the increasing rate is comparatively small in the next 40 min.

182 3.2 Viscosity prediction

183 3.2.1 Methodology for predicting the viscosity

184 For rheological properties of cement suspension, the volume fraction of particles is a major
185 determinant affecting its viscosity. It is generally considered that there is certain functional relation
186 between suspension viscosity and particle volume fraction [29]. PC-CSA paste can be regarded as
187 a suspension incorporating PC and CSA particles. Due to the differences in physicochemical
188 properties of PC and CSA, their viscosity characteristics before and after mixing together vary
189 significantly. Therefore, the viscosity of composite paste should be characterized by the volume
190 fraction of PC particles ($\varphi_{C/W}$) and the volume fraction of CSA particles (φ_S) respectively. The basic
191 prediction function is shown in Eq. (4):

$$192 \quad \eta = \eta_0 \cdot f(\varphi_{C/W}, \varphi_S) \quad (4)$$

193 where η is the viscosity of composite paste, η_0 is the viscosity of suspension medium (water).

194 Although the above function expression gives the method to predict viscosity by using $\varphi_{C/W}$ and φ_S

195 respectively, the result of one-time analysis of multivariate function expression is generally not

196 accurate enough. In this study, the volume fraction of PC is larger than that of CSA, so the composite
197 paste can be regarded as CSA suspended in PC-H₂O system. Therefore, the Eq. (4) can be rewritten
198 as Eq. (5) and Eq. (6):

$$199 \quad \eta = \eta_C \cdot f(\varphi_S) \quad (5)$$

$$200 \quad \eta_C = \eta_0 \cdot f(\varphi_{C/W}) \quad (6)$$

201 where η_C is the viscosity of PC suspension. According to Eq. (5) and Eq. (6), the prediction of
202 PC-CSA pastes viscosity can be divided into two steps: (1) $f(\varphi_{C/W})$ is predicted from the viscosity
203 of pure cement paste and water. (2) $f(\varphi_S)$ is predicted from the viscosity of PC-CSA pastes and
204 cement paste with the same PC volume fraction. Furthermore, according to the viscosity test results
205 of **Fig. 6**, CSA particles have greater impact on viscosity than PC particles. Therefore, the viscosity
206 prediction parameter $f(\varphi_{C/W})$ of PC with the characteristics of small viscosity influence and large
207 volume fraction at 5 min and 60 min was first calculated, and then the CSA viscosity parameter $f(\varphi_S)$
208 was computed.

209 **3.2.2 Prediction of $f(\varphi_{C/W})$**

210 In this study, the PC particle volume fractions of all pastes are more than 45%, belonging to high
211 concentration suspension. Krieger and Dougherty [30] have proposed a model which can well
212 predict the viscosity of various high concentration suspensions, and the equation is as follows:

$$213 \quad \eta_r = \left(1 - \frac{\varphi}{\varphi_m}\right)^{-K} \quad (7)$$

214 Where η_r is the relative viscosity (in this part, $f(\varphi_{C/W})$). φ and φ_m is the actual volume fraction of
215 particles and maximum particle packing density respectively. K is a fitting parameter. It can be found
216 from Eq. (7) that φ_m and K are two unknown parameters. Liu [31] found that the linear relationship
217 between $1 - \eta_r^{-1/2}$ and φ can be obtained by transforming the above expression when $K=2$. Based on

218 the above consideration, Liu modified the above function expression and added an adjustment
 219 coefficient. The modified expression is as follows:

$$220 \quad \eta_r = \left[b \left(1 - \frac{\varphi}{\varphi_m} \right) \right]^{-2} = [a(\varphi_m - \varphi)]^{-2} \quad (8)$$

221 Where $a=b/\varphi_m$. Rearranging the terms in Eq. (8), leads to Eq. (9) as:

$$222 \quad 1 - \eta_r^{-1/2} = 1 - a\varphi_m + a\varphi \quad (9)$$

223 **Fig. 9** shows the value of a and φ_m calculated by Eq. (9) when the correlation coefficient R^2 reach
 224 the maximum. It can be seen in **Fig. 9** that R^2 is greater than 0.8 in both 5 min and 60 min, indicating
 225 the fitting accuracy is relatively high. Therefore, it is feasible to use Eq. (9) to predict $f(\varphi_{C/W})$. The
 226 prediction functions of PC paste at 5 min and 60 min are shown in Eq. (10) and Eq. (11):

227 5 min:

$$228 \quad f(\varphi_{C/W}) = [0.097 * (0.9144 - \varphi_{C/W})]^{-2} \quad (10)$$

229 60 min:

$$230 \quad f(\varphi_{C/W}) = [0.0689 * (1.0377 - \varphi_C)]^{-2} \quad (11)$$

231

232 3.2.3 Prediction of $f(\varphi_s)$

233 Compared with PC, the volume fraction of CSA particle is much smaller and the values of φ_s in
 234 this study are no more than 20%. For the suspension system with relatively small concentration,
 235 some models can be used to predict the viscosity by volume fraction. Among them, one of the widely
 236 used models is Einstein model [32]:

$$237 \quad \eta_r = \frac{\eta}{\eta_0} = 1 + 2.5\varphi \quad (12)$$

238 Where η_r is the relative viscosity (the ratio of the suspension viscosity to the dispersion medium

239 viscosity). However, this model is basically applicable to the case where φ tends to zero. There are
240 other models that can be used to calculate the relative viscosity by particle volume fraction. The
241 two most frequent models are Batchelor model [33] and Brinkman model [34], which is shown in
242 Eq. (13) and Eq. (14):

$$243 \quad \eta_r = 1 + 2.5\varphi + 6.2\varphi^2 \quad (13)$$

$$244 \quad \eta_r = \frac{1}{(1-\varphi)^{2.5}} \quad (14)$$

245 Among them, Batchelor model is based on Einstein model and takes Brownian motion into
246 account. Meanwhile, since the Brinkman model is derived by adding more particles into the diluted
247 suspension to increase the viscosity, it can be applied to suspensions with higher concentration than
248 Einstein model. When φ is very small, Brinkman model can be simplified to Eq. (15):

$$249 \quad \eta_r = \frac{1}{1-2.5\varphi} \quad (15)$$

250 All the above models can predict the viscosity by the volume fraction of suspension particles.
251 However, these models do not consider the effect of particles water film thickness in the suspension
252 [35-37]. The existence of adsorbed water layer will make the actual volume fraction of particles in
253 the suspension larger than the calculated particle volume fraction. Assuming that the particles in the
254 suspension are spherical, the actual volume fraction k can be obtained by Eq. (16):

$$255 \quad k = \left(1 + \frac{r}{R}\right)^3 \quad (16)$$

256 Where R and r are the diameters of suspension particle and water film thickness respectively.
257 Considering the influence of adsorbed water layer, an unknown parameter k can be added to the
258 above viscosity prediction model:

$$259 \quad \text{Einstein model: } \eta_r = (1 + 2.5k\varphi) \quad (17)$$

260 Batchelor model: $\eta_r = 1 + 2.5k\varphi + 6.2(k\varphi)^2$ (18)

261 Brinkman model: $\eta_r = \frac{1}{(1-k\varphi)^{2.5}}$ (19)

262 Simplified Brinkman model: $\eta_r = \frac{1}{1-2.5k\varphi}$ (20)

263 These four models are adopted to predict the $f(\varphi_s)$ of composite pastes with CSA as solute. The
 264 relative viscosity of composite paste with different PC and CSA contents are shown in **Table 4** and
 265 **Table 5**. From the transverse view, when the φ_s is the same, the relative viscosity of different PC
 266 volume fraction pastes has little difference. In other words, in the prediction of $f(\varphi_s)$, the value of
 267 $f(\varphi_s)$ is only related to φ_s , but not to $\varphi_{C/W}$. This is consistent with the original assumption in
 268 predicting viscosity that firstly predict the viscosity of PC-H₂O system, and then predict the
 269 viscosity of PC-CSA-H₂O paste. The average relative viscosity of 5 groups under the same φ_s is
 270 taken as $f(\varphi_s)$, and the above four models are used to fit the relationship between $f(\varphi_s)$ and φ_s (shown
 271 in **Fig. 10**). It can be seen that whether at 5 min or 60 min, Einstein model can well characterize the
 272 relationship between $f(\varphi_s)$ and φ_s when taking into account the water film thickness. Based on the
 273 original data, the relative viscosity at 5 min and 60 min can be predicted by Eq. (21) and Eq. (22):

274 5 min:

275 $f(\varphi_s) = 1 + 5.1288 * \varphi_s$ (21)

276 60 min:

277 $f(\varphi_s) = 1 + 6.8008 * \varphi_s$ (22)

278 3.2.4 Prediction results and accuracy

279 After getting the function expression of $f(\varphi_{C/W})$ and $f(\varphi_s)$, the prediction model of η_{\min} can be
 280 calculated. By substituting the expressions of $f(\varphi_{C/W})$ and $f(\varphi_s)$ at different resting time into Eq. (5)

281 and Eq. (6), the viscosity prediction equations of PC-CSA pastes at 5 min and 60 min are obtained

282 respectively:

283 5 min:

$$284 \quad \eta = 0.000936 * [(0.097 * (0.9144 - \varphi_{C/W})]^{-2} * (1 + 5.1288 * \varphi_S) \quad (23)$$

285 60 min:

$$286 \quad \eta = 0.000936 * [(0.0689 * (1.0377 - \varphi_{C/W})]^{-2} * (1 + 6.8008 * \varphi_S) \quad (24)$$

287 In order to evaluate the prediction results of the above two equations, R^2 is used here to
288 characterize the accuracy as well. As can be seen in **Fig. 11**, the R^2 of 60 min is lower than that of
289 5 min, indicating the prediction accuracy of η_{\min} at 5 min is higher than that at 60 min. Based on
290 the experimental results of viscoelasticity, it can be found that there are large differences among
291 the values of G' , G'' and δ at 5 min and 60 min, showing their colloidal interactions of particles
292 and microstructure are different. Yield stress and viscosity are two common rheological
293 parameters for fresh cement paste. The yield stress of cement paste comes from the microscopic
294 network structure formed by colloidal interaction [38, 39]. When the cement paste is in a static
295 state, colloidal flocculation and cement hydration bond (i.e. C-S-H bridge) will form
296 microstructure in paste, thus increasing the yield stress [39]. Viscosity refers to the resistance to
297 flow and it can characterize the deformation speed of cement paste. Unlike yield stress, viscosity
298 is more related to energy dissipation during flow behavior [40], i.e., volume fraction of particles
299 [41, 42], Specific surface area [43], packing effect [44] and friction effect [45]. With the
300 extension of resting time, the viscoelastic properties and factors such as flocculation network
301 structure and colloidal attractive forces related to yield stress of cement mixes change. These
302 variations do not significantly influence the relationship between particle volume fractions and

303 viscosity at different resting time (As can be seen from Eq. (23) and Eq. (24), the form of two
304 equations is still the same and the parameters are close to each other), but decrease the R^2 from
305 0.9794 to 0.9639. The change of viscoelasticity at 5 min and 60 min due to the formation of
306 flocculation structure and hydration products is the main reason for the decline of viscosity
307 prediction accuracy. Therefore, it is reasonable to believe that an appropriate prediction time is 5
308 min if the particles volume fraction is used to predict the η_{\min} of PC-CSA pastes.

309 **4 Conclusions**

310 (1) With the increase of shear rate, PC-CSA pastes under different mix proportions show shear
311 thinning behavior first and then shear thickening, and there is a η_{\min} in each paste during the whole
312 shear process. Higher PC and CSA particles volume fraction cause a larger η_{\min} .

313 (2) All PC-CSA pastes have linear viscoelastic regions. When G' or G'' is less than critical strain,
314 they can keep constant in a certain range. With the increase of $f(\varphi_{C/W})$ and $f(\varphi_S)$, the value of G' and
315 G'' corresponding to the critical strain increases, and the critical strain and flow point strain also
316 increase. With the prolongation of resting time, the phase angles of composite pastes change
317 significantly at 5min and 60 min, demonstrating that the viscoelasticity is in a trend of dynamic
318 variation.

319 (3) Einstein model and Krieger and Dougherty model are used to describe the functional
320 expressions of $f(\varphi_{C/W})$ and $f(\varphi_S)$ respectively. The relationships between η_{\min} , $\varphi_{C/W}$ and φ_S at 5 min
321 and 60 min are obtained under the condition of considering the water film thickness. The R^2 values
322 of the prediction models at two resting time are both high, which proves the accuracy of calculated
323 results. By comparing the results it can be found that the prediction model at 5 min has higher
324 accuracy than that of 60 min. Besides the particles volume fraction, the formation of flocs and

325 hydrated products also have a certain impact on the rheological properties with the extension of
326 resting time.

327 **Declaration of competing interest**

328 The authors declare that they have no known competing financial interests or personal
329 relationships that could have appeared to influence the work reported in this paper.

330 **Acknowledgements**

331 This work was financially supported by China Scholarship Council (grant number:
332 202006370082).

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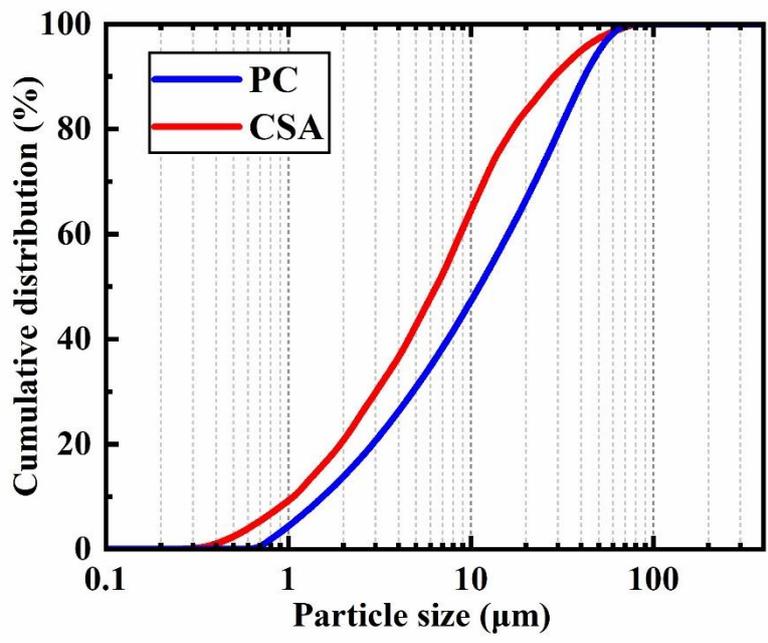
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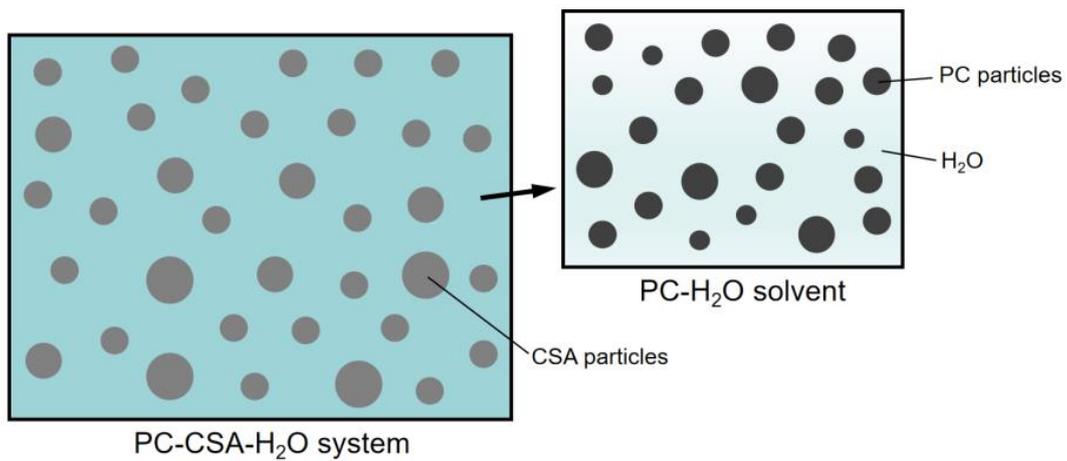


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Fig. 1 Particle size distributions of PC and CSA

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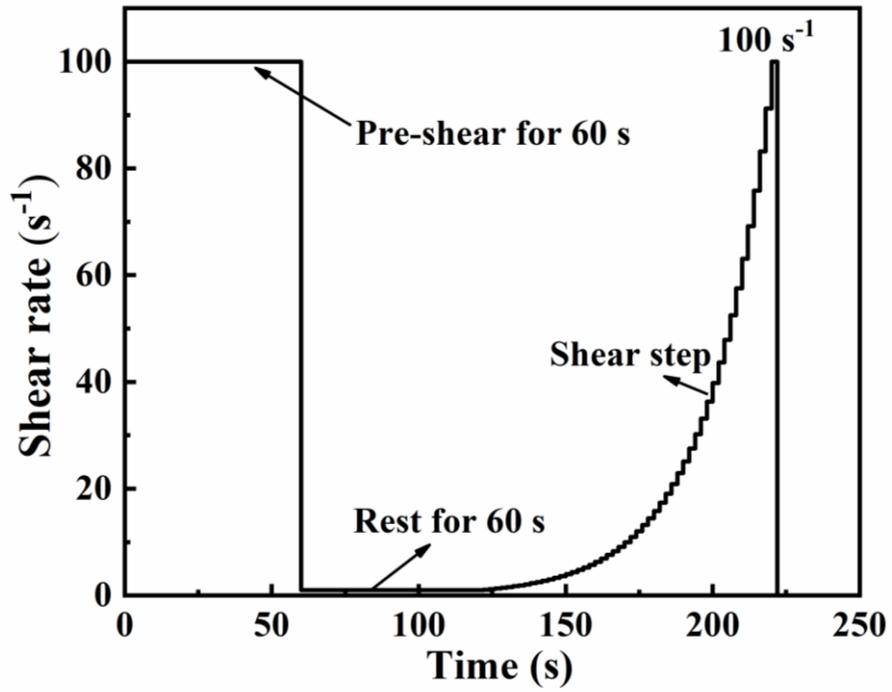


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Fig. 2 An illustration of PC-CSA-H₂O system

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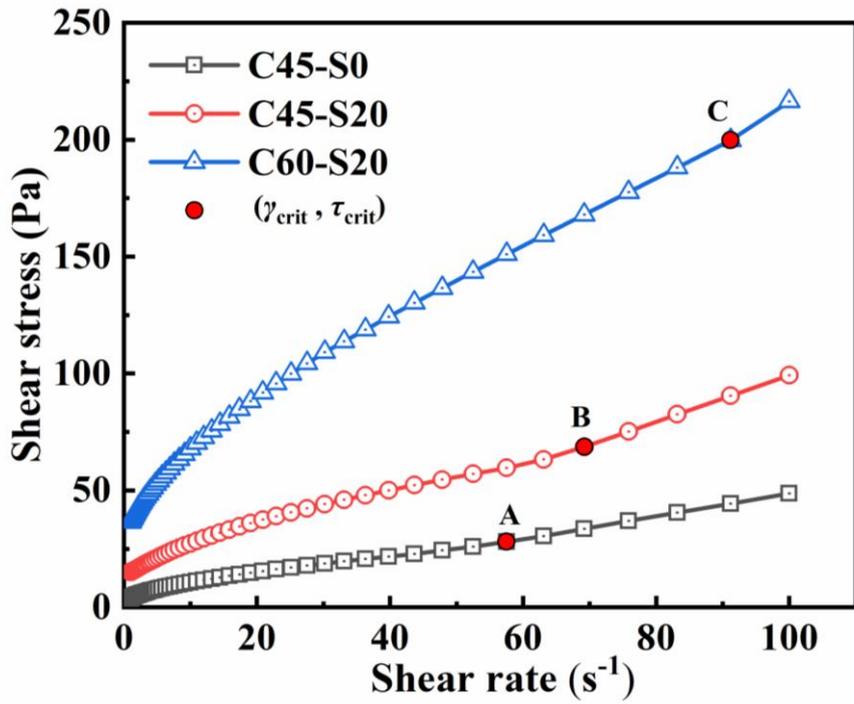


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Fig. 3 Test protocol for dynamic rheological test

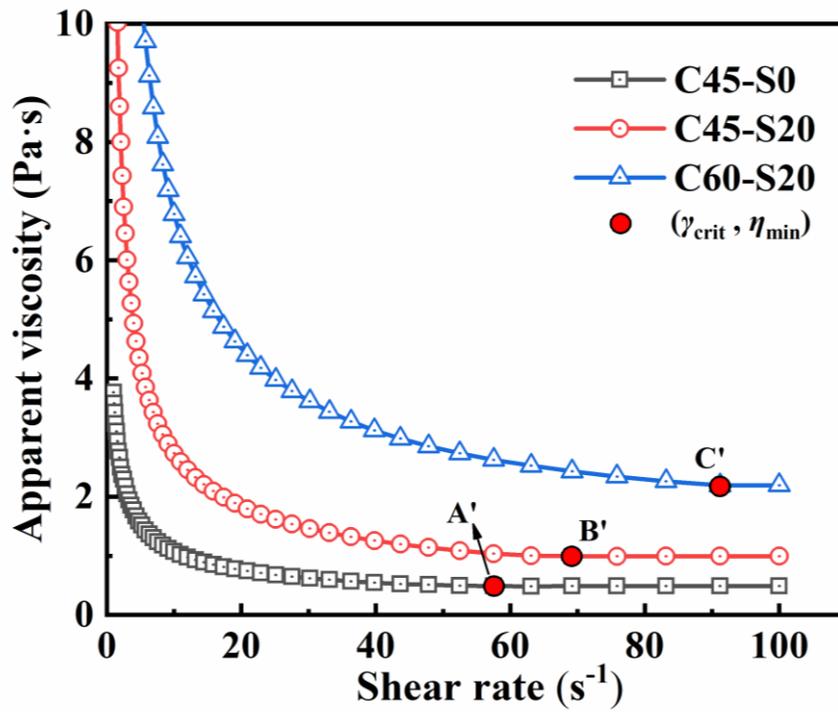
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(a) Shear rate vs. shear stress



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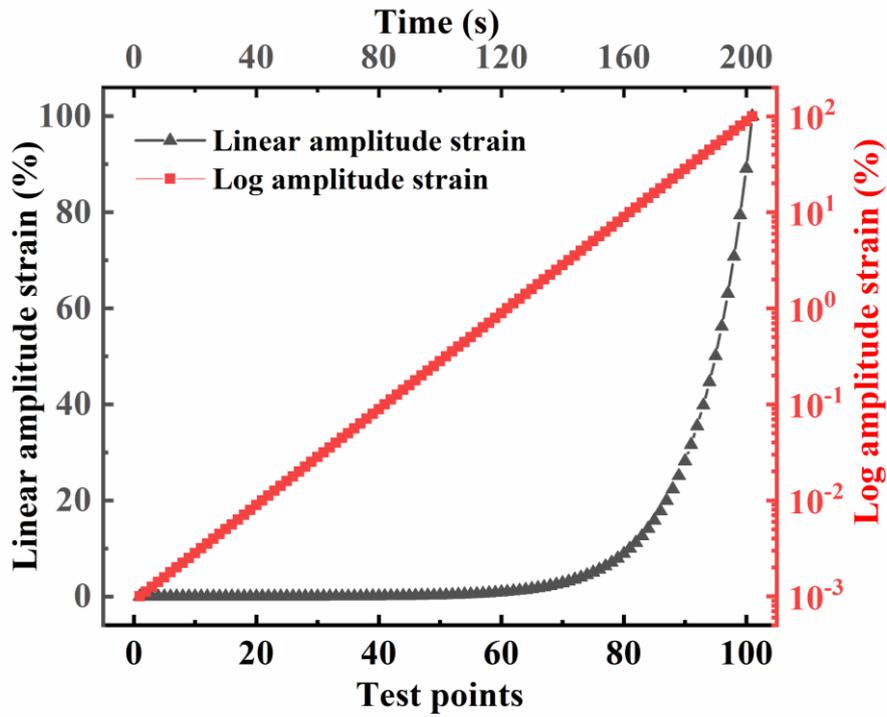
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(b) Shear rate vs. apparent viscosity

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Fig. 4 Rheological curves of PC-CSA pastes

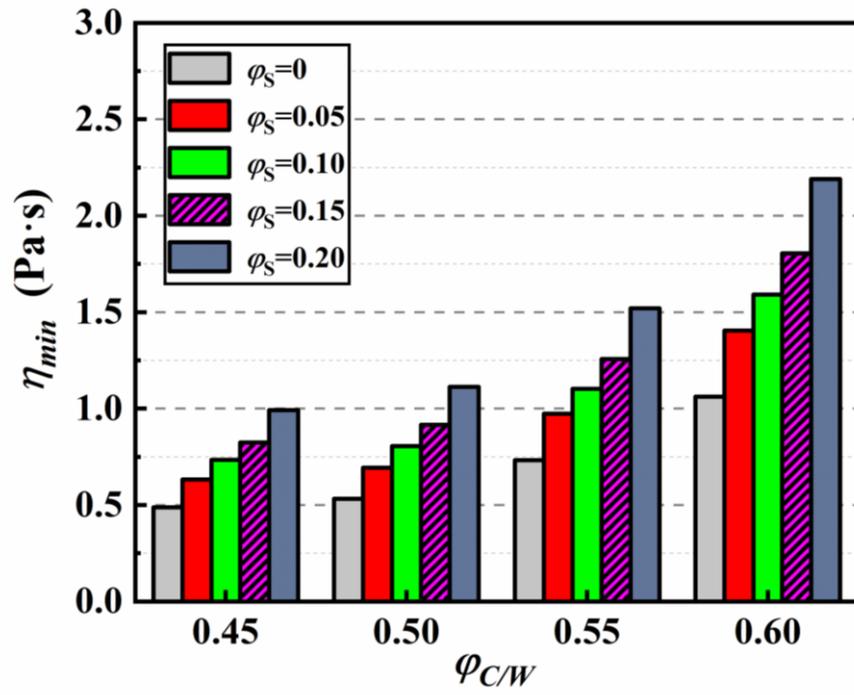
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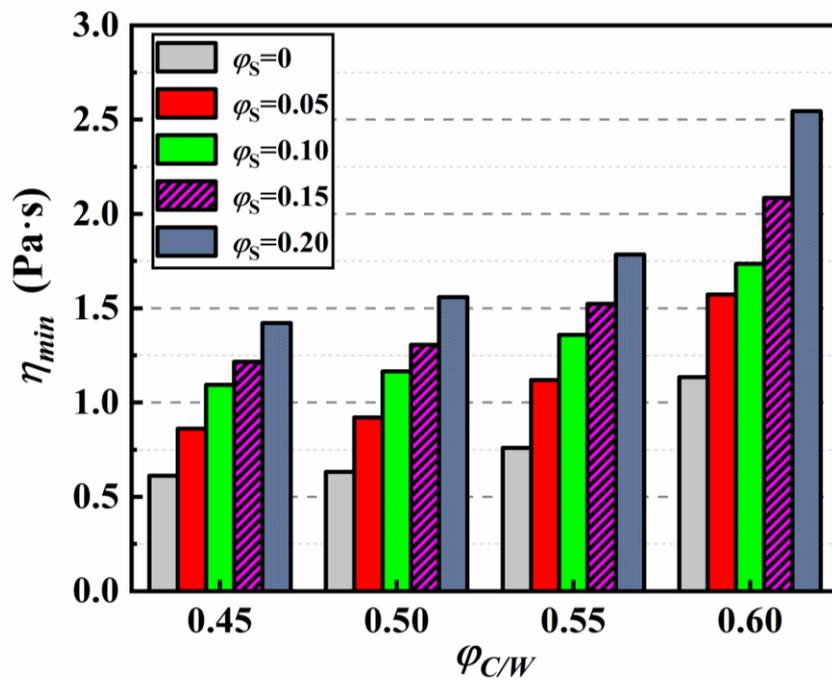
Fig. 5 Test protocol for amplitude sweep



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(a) 5 min



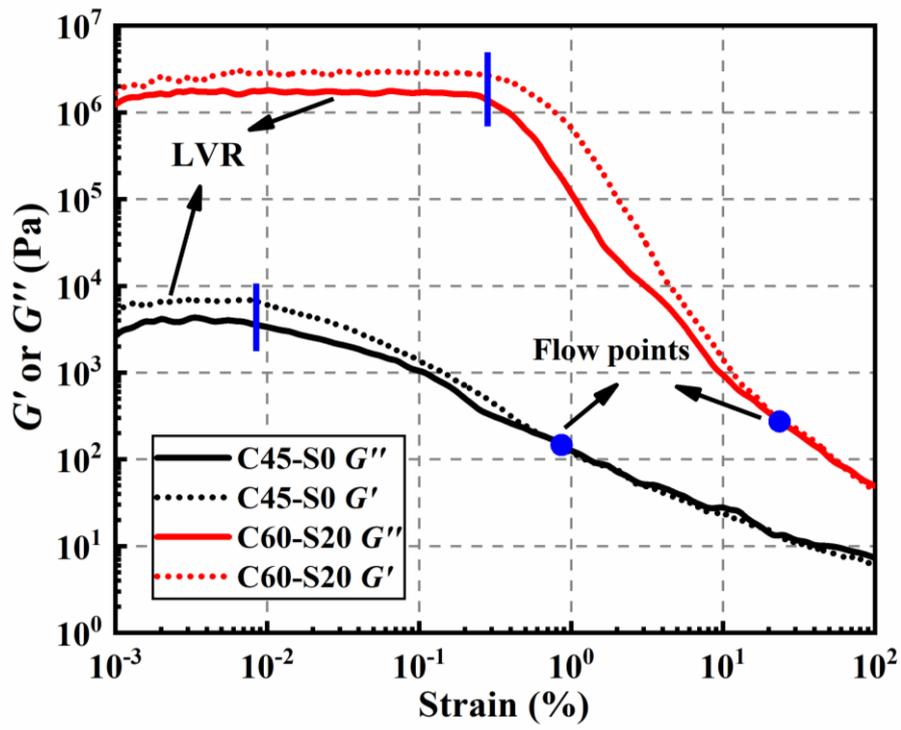
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(b) 60 min

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Fig. 6 The minimum apparent viscosities of PC-CSA pastes



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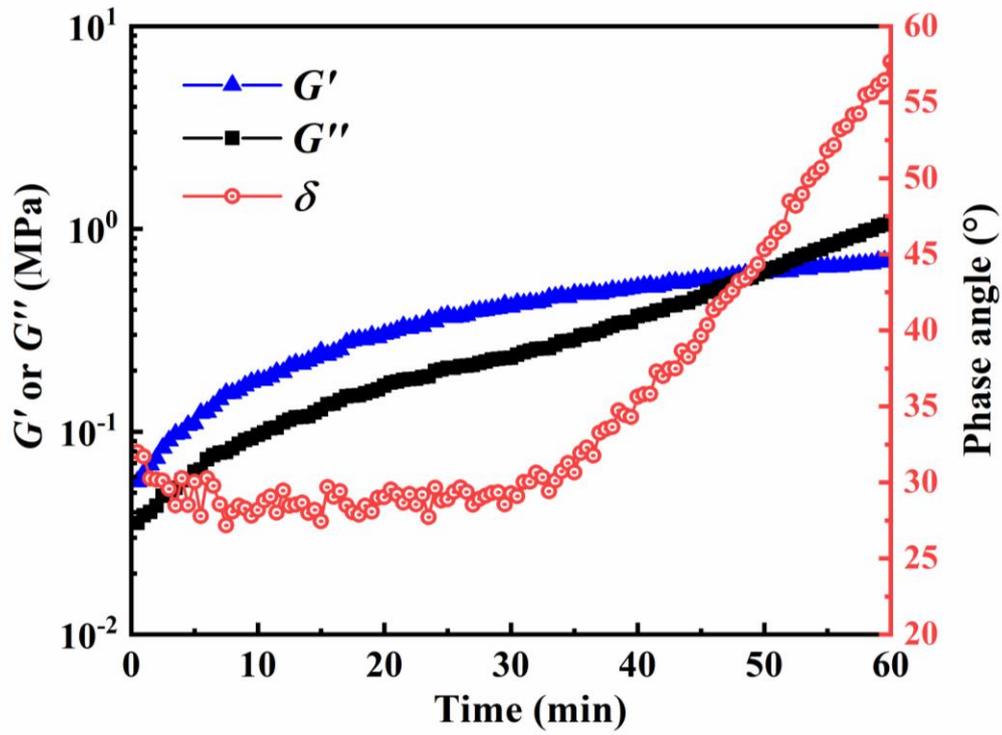
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Fig. 7 Typical amplitude sweep curves of PC-CSA pastes

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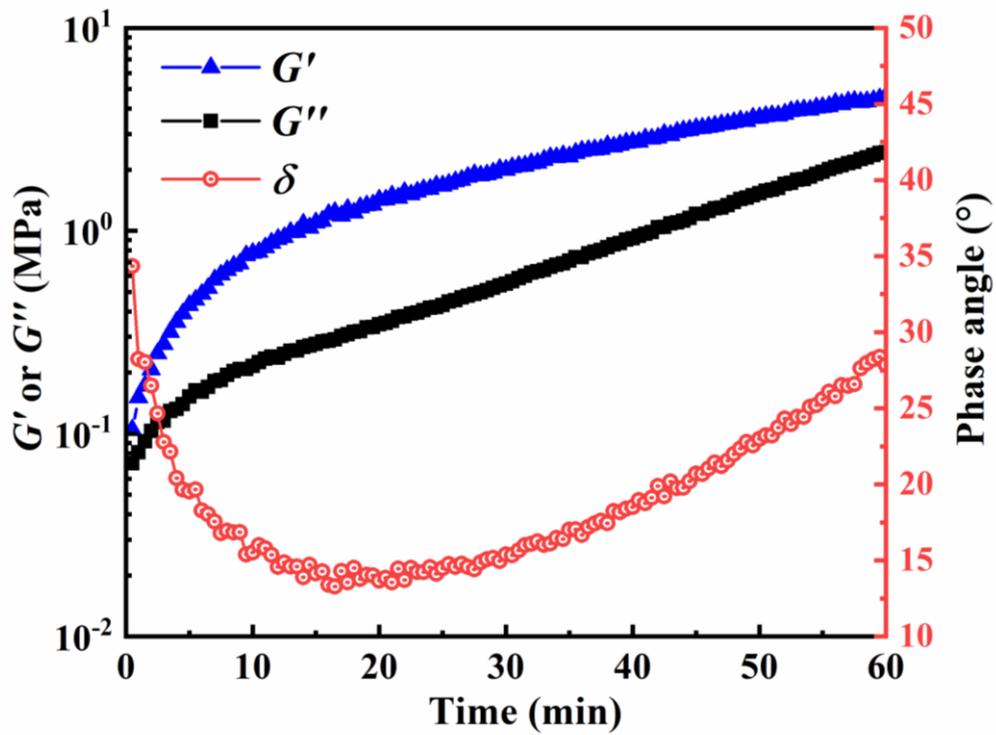
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(a) C45-S0



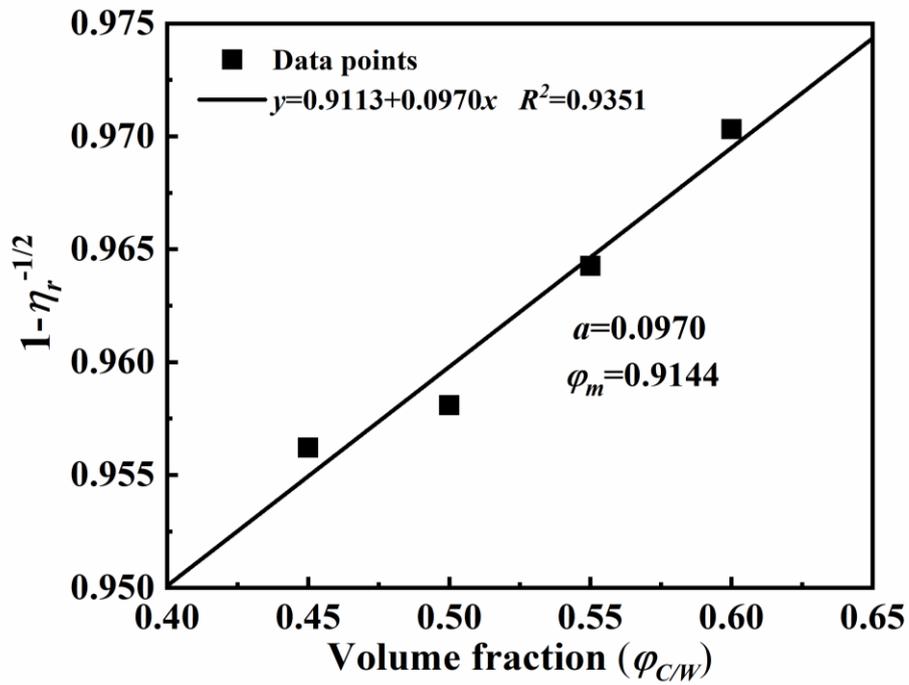
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(b) C45-S20

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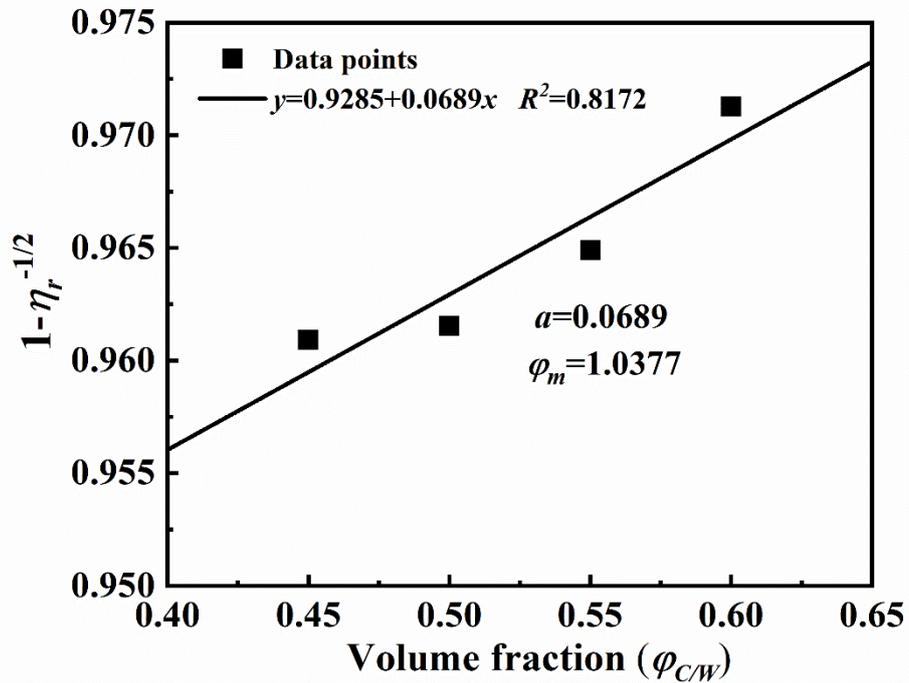
Fig. 8 Evolution of G' , G'' and δ with time at a frequency of 1 Hz and a strain amplitude of 10^{-4}



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(a) 5min



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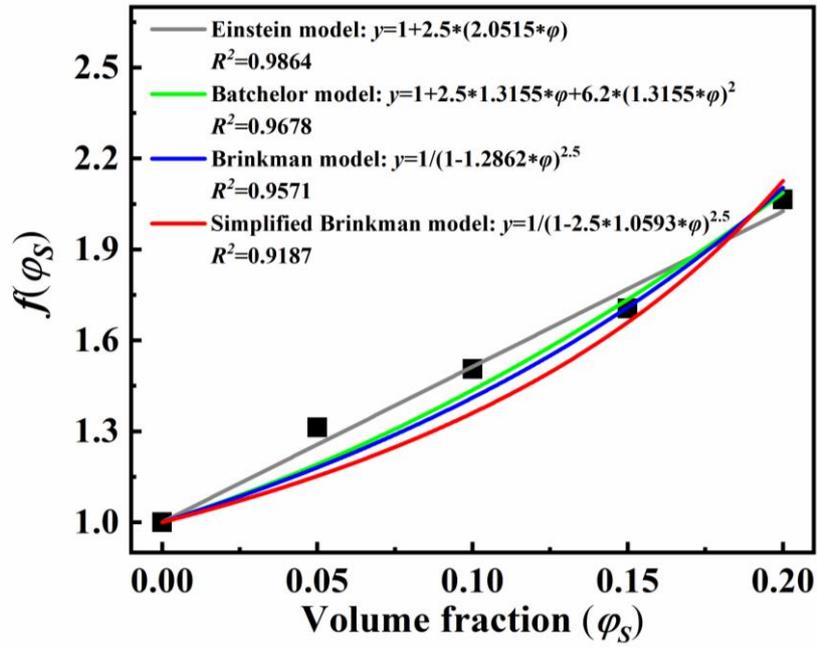
502

(b) 60 min

503 **Fig. 9** The values of a and φ_m calculated by the linear relationship between $1-\eta_r^{-1/2}$ and φ when R^2

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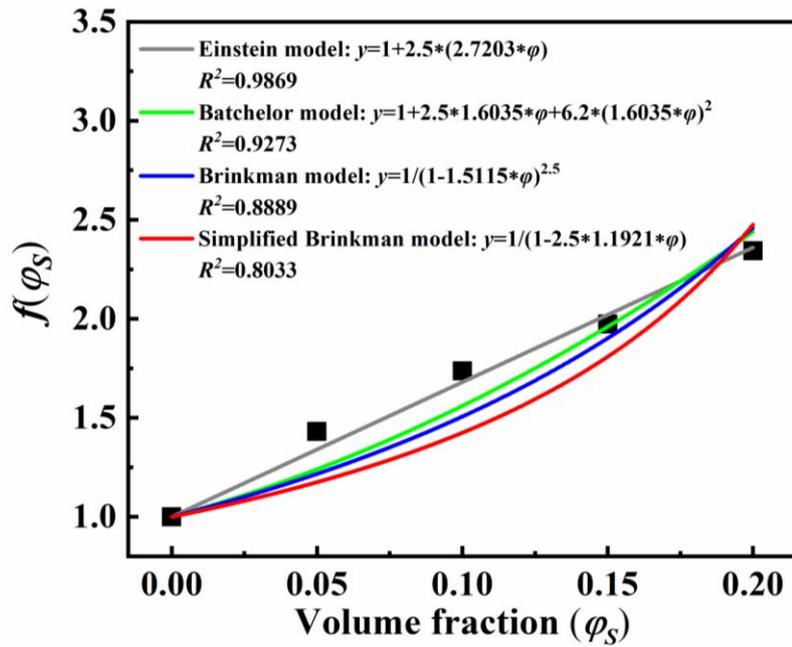
reaches the maximum



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(a) 5 min



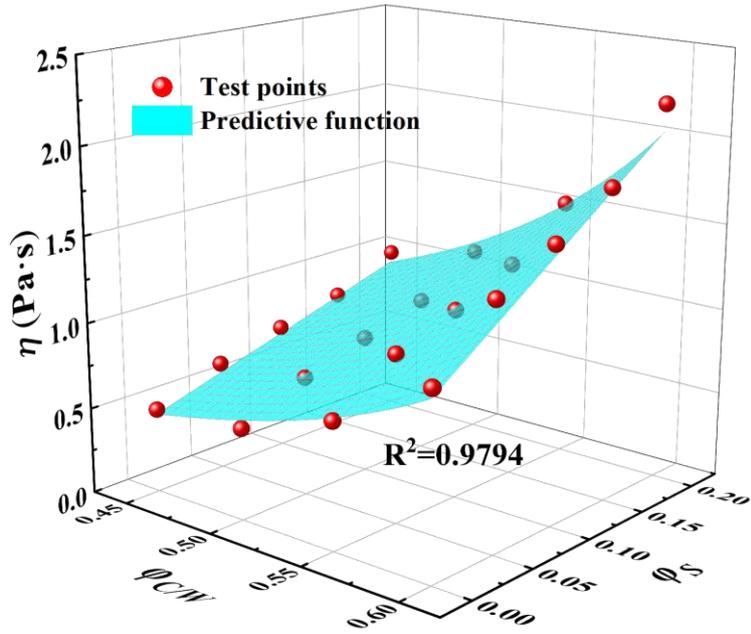
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(b) 60 min

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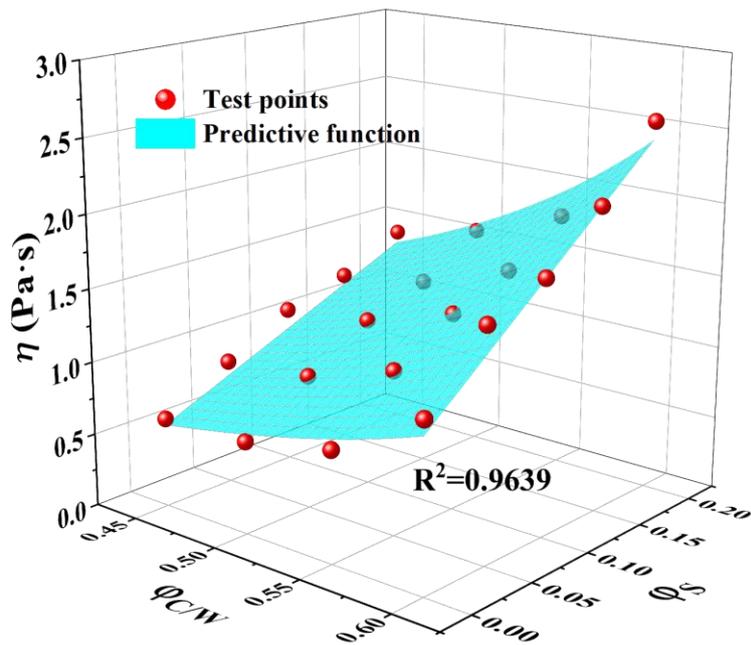
Fig. 10 The relationships between $f(\varphi_s)$ and φ_s fitted by different prediction models



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(a) 5 min



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(b) 60 min

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Fig. 11 The prediction results of PC-CSA pastes at different resting time

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Table 1 Physical properties and chemical compositions of PC and CSA

Raw materials	Composition (wt.%)						Ignition loss (%)	Density (g/cm ³)
	CaO	Al ₂ O ₃	SiO ₂	Fe ₂ O ₃	SO ₃	MgO		
PC	62.26	7.31	20.76	3.25	2.81	2.91	2.10	3.15
CSA	46.79	22.83	11.72	1.50	12.09	1.72	1.33	2.80

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Table 2 Mixture proportions of PC-CSA systems

Sample	Volume ratio (%)			Mass ratio (%)			W/B
	$\varphi_c + \varphi_w$	$\varphi_c / (\varphi_c + \varphi_w)$	φ_s	PC	CSA	H ₂ O	
C45-S0	100	45	0	141.8	0	55	0.388
C45-S5	95	45	5	134.7	14	52.3	0.351
C45-S10	90	45	10	127.6	28	49.5	0.318
C45-S15	85	45	15	120.5	42	46.8	0.288
C45-S20	80	45	20	113.4	56	44	0.260
C50-S0	100	50	0	157.5	0	50	0.317
C50-S5	95	50	5	149.6	14	47.5	0.290
C50-S10	90	50	10	141.8	28	45	0.265
C50-S15	85	50	15	133.9	42	42.5	0.242
C50-S20	80	50	20	126	56	40	0.220
C55-S0	100	55	0	173.3	0	45	0.260
C55-S5	95	55	5	164.6	14	42.8	0.239
C55-S10	90	55	10	155.9	28	40.5	0.220
C55-S15	85	55	15	147.3	42	38.3	0.202
C55-S20	80	55	20	138.6	56	36	0.185
C60-S0	100	60	0	189	0	40	0.212
C60-S5	95	60	5	179.6	14	38	0.196
C60-S10	90	60	10	170.1	28	36	0.182
C60-S15	85	60	15	160.7	42	34	0.168
C60-S20	80	60	20	151.2	56	32	0.154

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Table 3 Critical strain, G' , G'' and flow point strain

Sample	Critical strain (%)	G' at critical strain (Pa)	G'' at critical strain (Pa)	Flow point strain (%)
C45-S0	0.0079	6,902	4,026	0.7943
C45-S20	0.0158	82,190	63,910	2.512
C60-S0	0.0251	315,800	228,200	3.545
C60-S20	0.1995	2,812,000	1,619,000	19.95

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Table 4 The relative viscosity of PC-CSA paste at 5 min ($f(\varphi_s)$)

φ_s	$\varphi_{C/W}$				Average value
	45	50	55	60	
0	1.000	1.000	1.000	1.000	1.000
5	1.298	1.302	1.328	1.323	1.312
10	1.504	1.512	1.505	1.498	1.505
15	1.688	1.718	1.716	1.699	1.705
20	2.032	2.088	2.074	2.061	2.064

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Table 5 The relative viscosity of PC-CSA paste at 60 min ($f(\varphi_s)$)

φ_s	$\varphi_{C/W}$				Average value
	45	50	55	60	
0	1.000	1.000	1.000	1.000	1.000
5	1.407	1.454	1.474	1.386	1.431
10	1.785	1.841	1.788	1.530	1.736
15	1.987	2.065	2.005	1.838	1.974
20	2.321	2.462	2.350	2.244	2.344

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