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Incorporation of MnO₂ Nanoparticles into MOF-5 for Efficient Oxygen Evolution Reaction

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MOF-5 MnO2 MoF-5 MNO2 MNO2 MNO2 MNO2 MNO2 MNO2 M

Graphical Abstract

Abstract

A composite MnO₂@MOF-5 is prepared by in-situ incorporation of pre-synthesized MnO₂ nanoparticles into metal organic framework, MOF-5, during synthesis. The product is characterized by Powder X-Ray diffraction analysis, Raman spectroscopy, Fourier transform infrared spectroscopy, Ultraviolet-Visible spectroscopy, Scanning Electron Microscopy, Energy Dispersive X-ray spectroscopy and elemental mapping, which support the formation of proposed composite materials. The oxygen evolution reaction activity of MnO₂@MOF-5 composite is evaluated by cyclic voltammetry, linear sweep voltammetry and Chronoamperometric measurement under visible light. It is found that MnO₂@MOF-5/NF has better durability and ability to produce current density of 10 mAcm⁻² at only 324 mV overpotential with lower 71 mVdec⁻¹ Tafel slope as compared to some of previously reported Mn-based catalysts for OER. Stability of these electrodes is evaluated by Chronoamperometric studies for 6000 seconds in presence of visible light and they showed constant current density. Furthermore, stability studied by continuous CV sweeps in 1.0 M NaOH at a scan rate of 100 mVs⁻¹ shows that these materials are stable upto 100 cycles, which confirms the stability and durability of the electrodes.

Key Words: MOF-5, MnO₂ nanoparticles, in-situ incorporation, Linear sweep voltammetry, Oxygen evolution reaction

Introduction

Increasing environmental pollution and rapid decrease of fossil fuels have diverted the attention of researchers towards the development of new alternate sources of energy [1]. Over the past few decades, many efforts have been made in this regard and it has been found that hydrogen can be used as a clean, secure, non-polluting and environment friendly source of energy [2]. Hydrogen can be produced from the hydrogen containing compounds like fossil fuels and biomass but they contaminate the environment [3]. Alternately, hydrogen can be effectively produced by photocatalytic water splitting which has no harmful effect upon environment [4]. The production of hydrogen by water splitting has attracted great attention after first reported by Fujishima and Honda in 1972 [5].

During water splitting, hydrogen and oxygen produced at surface of catalyst due to redox reaction [6]. Currently, Ru/Ir-based OER based catalysts have been emerged as best OER catalysts but due to their low earth abundant and high cost, they cannot be used at commercial levels [7-8]. So, development of highly active OER catalyst based on low cost earth abundant 3d-element has attracted a huge research attention. Therefore, up till now, more than hundred different materials have been reported as catalyst for water splitting [9]. However, high electron-hole pair recombination lowers the solar energy conversion and reduces the efficiency of catalysts [10]. Many efforts have been made to solve this problem but efficiency has not been increased up to satisfactory level [11].

In the recent years, metal organic frameworks (MOFs) have emerged as potential candidate for different applications such as storage [12], separation [13], catalysis [14] and biological imaging [15]. MOFs have unique properties such as high surface area and large number of cavities and channels [16]. In 2010, Garcia and co-workers reported Zr containing MOFs known as UiO-66 and UiO-66 (NH₂) as prominent catalysts for water splitting in the presence of visible light [17]. It has been observed that both UiO-66 and UiO-66 (NH₂) have greater catalytic activity towards water splitting when Pt nanoparticles are incorporated within the pores [18]. Since then many successful efforts have been made for preparation of composites of MOFs by incorporation of different materials such as metal/metal oxide nanoparticles [19], quantum dots [20], graphene [21], dense and porous silica nanospheres [22] and magnetic beads [23] to get specific properties. In MOFs, the organic ligands serve as antenna to harvest light and activate the metal. Photogenerated

electrons produced in MOFs are transferred to nanoparticles and increase the charge separation for efficient photocatalytic activity [24].

Manganese-based oxides (MnO_x) nanomaterials have been emerged as efficient OER catalysts due to their unique properties such as well-controlled morphology and electronic state [25-26]. Manganese oxide (MnO₂) nanoparticles, due to their stability, large surface area and small size [27] are used in redox reactions [28] and catalysis [29]. MOF-5 is an important metal organic framework that consists of $[Zn_4O]^{6+}$ clusters linked by octahedral arrangement of 1,4-Benzenedicarboxylate groups to form a cubic porous framework of MOF-5[30]. In the present work, nanoparticles of MnO₂ have incorporated into MOF-5 to form MnO₂@MOF-5 composite. Different experimental conditions are tested and composite materials showed superior OER performance to individual MnO₂ particles.

Experimental

Chemicals

The chemicals used for the synthesis of materials were zinc acetate dihydrate, Zn(CH₃COO)₂.2H₂O, 1,4-benzenedicarboxylic acid (H₂BDC), triethylamine (TEA), N,N-dimethylformamide (DMF), KMnO₄, MnSO₄.H₂O and H₂O₂. All these chemicals were purchased from Merck. These were of analytical grade and used as such without any further purification.

Synthesis of MnO₂@MOF-5 Composite

Hydrothermal method was used for the preparation of nanoparticles of MnO₂ as reported in literature [31]. MnO₂@MOF-5 composite was prepared by incorporation of pre-synthesized nanoparticles of MnO₂ into MOF-5 during its synthesis. In a typical procedure, a suspension of 10 mg MnO₂ nanoparticles was prepared in 20 mL DMF. The suspension was added into 100 mL DMF solution containing 2.7g Zn(CH₃COO)₂.2H₂O. Then 50 mL DMF solution containing 0.8 g H₂BDC and 2 mL TEA was added dropwise to the mixture under constant magnetic stirring. The whole mixture was stirred at room temperature for 24 hours. The brownish precipitates were obtained and collected by centrifugation, washed with distilled water and DMF several times. The resulting product was dried at 50°C in vacuum oven for 3 hours and activated at 120°C for 6 hours. The dried precipitates were ground and stored for further characterization and photoelectrochemical studies. A sample of pure MOF-5 was also prepared by the same procedure without adding MnO₂ nanoparticles.

Oxygen evolution reaction studies

The oxygen evolution reaction studies of synthesized samples were studied in presence of visible light as well as in dark at room temperature by Cyclic voltammetry (CV), Linear sweep voltammetry (LSV) and Chronoamperometry using the electrochemical workstation (Autolab PG station 204) in 1.0 M NaOH aqueous electrolyte at various scan rates. The electrochemical measurements were conducted using a three-electrode setup containing Ag/AgCl as reference electrode and Pt-wire as counter electrode. All the applied potentials were converted into reversible hydrogen electrode (RHE) by using following equation

$$E_{RHE} = E_{Ag/AgC1/Sat. KC1} + 0.059 \text{ pH} + 0.197 \tag{1}$$

LSV was measured at 1 mVs⁻¹ scan rate and it was used to evaluate Tafel plot according to following equation

$$\eta = a + b \log j \tag{2}$$

Where η is overpotential, j is current density and b is the Tafel slope. For OER overpotential was calculated by following equation

$$\eta = E_{\rm RHE} - 1.23 \tag{3}$$

The working electrode was prepared on nickel-foam (NF). For the preparation of working electrode, a piece of NF (1cm \times 1cm) was cleaned with ethanol and acetone by sonication for 30 minutes, respectively, washed with distilled water, and dried at room temperature. 10 mg of the prepared sample was added in distilled water to make the slurry, which was then uniformly pasted on NF and dried at 50°C for overnight. CV, LSV and Chronoamperometric measurements used the dried NF containing sample as working electrode for the study of water splitting activity. It was observed from SEM images that catalytic material remains deposited on the surface of Ni foam before and after photoelectrochemical reaction (Figure 1). It indicated the significant stability of these working electrodes for PEC studies.



Figure 1: Schematic representation of coating on Ni-Foam and its use for photoelectrochemical study; inset are the magnify images of blank Ni-foam, Working electrode before CV studies and after CV studies

Characterization

Powder X-ray diffraction (XRD) patterns of pure MOF-5 and MnO₂@MOF-5 were recorded on a Shimadzu XRD diffractometer with Cu-K α radiation (λ =0.15406nm) in the range of 2 θ between 10° to 80° at scan rate of 5 degree min⁻¹. Raman spectrometry was carried by using Horiba Jobin-Yvon Lab RAM HR800 Raman Spectrometer in range from 100 to 2000 cm⁻¹. The 532 nm solid state laser was used to avoid florescence and decomposition of samples. Output power was reduced to 10% and acquisition time ranges from 5 to 10 minutes. Fourier transform infrared spectra were obtained by using Nicolet Nexus 870 in range from 4000-400 cm⁻¹. The morphology and composition of samples were studied by scanning electron microscopy (SEM) by using Philips XL30 Environmental SEM attached with Oxford Instrument Inca 500 Energy dispersive X-ray Spectrometer (EDX). The optical properties of the samples were studied by using Shimadzu UV-2600 UV-Visible spectrophotometer at room temperature in the range between 200 nm-900 nm.

Results and Discussion

Powder XRD patterns of MOF-5 and MnO₂@MOF-5 are shown in Figure 2, both the materials have grown in well crystalline form and intense peaks are observed. The P-XRD pattern of pure MOF-5 matches well with that reported in literature [32-33]. In the P-XRD pattern of MnO₂@MOF-5 the peaks that index to both MnO₂ and MOF-5 are observed and well defined diffraction peaks at 20 about 19.18°, 20.47°, 30.52° correspond to MOF-5 and 13.55°, 18.61°, 28.45°, 38.65°, 41.11°, 50.13° to MnO₂. The diffraction peaks correspond to MnO₂ matches with

the standard XRD pattern ICSD 44-141. From the P-XRD patterns of MOF-5 and MnO₂@MOF-5, it is observed that the host MOF-5 maintains its characteristics reflection pattern and crystallinity.



Figure 2: Powder XRD patterns of MOF-5 and MnO₂@MOF-5 in comparison with simulated pattern of MOF-5

The Raman spectra of MOF-5 and MnO₂@MOF-5 are shown in figure S1 (supporting information). The Raman spectrum of MOF-5 consists of five strong Raman bands at 635 cm⁻¹, 865 cm⁻¹, 1139 cm⁻¹, 1448 cm⁻¹ and 1615 cm⁻¹ and these are in accordance with the results reported in literature [34]. These Raman bands are due to vibrational modes of benzene rings and vibration modes of carboxylate groups [35]. Similarly, the Raman spectrum of MnO₂@MOF-5 also consist of five Raman bands at 635 cm⁻¹, 865 cm⁻¹, 1141 cm⁻¹, 1434 cm⁻¹ and 1615 cm⁻¹, which indicates that MOF-5 shows its dominance and maintain its crystalline structure. In Raman spectrum of MnO₂@MOF-5 slight Raman shifts can be observed at 1141 cm⁻¹ and 1434 cm⁻¹ which may be due to interaction of MnO₂ with [Zn₄O]⁶⁺ clusters of MOF-5 and some displacement of organic ligand.

These five vibrational modes of MOF-5 and MnO₂@MOF-5 are comparable with DFT quantum calculations of MOF-5as reported previously [36]. The Raman spectra of MnO₂@MOF-5 does not contain characteristic peaks of MnO₂ because of dominance of host MOF, peaks of MnO₂ are masked.

FTIR spectra of both MOF-5 and MnO₂@MOF-5 are shown in figure S2. The FTIR spectra of MOF-5 shows two sharp vibrational peaks at 1575 cm⁻¹ and 1370 cm⁻¹ due to symmetric and

asymmetric stretching vibration of C-O whereas $MnO_2@MOF-5$ shows vibrational bands at 1586 cm⁻¹ and 1377 cm⁻¹ due to symmetric and asymmetric stretching vibration of C-O bonded to Zn, respectively. In the FTIR spectra of MOF-5, symmetric stretching vibration of Zn₄O appears at 642 cm⁻¹ [37]. A strong peak appears at 530 cm⁻¹ in FTIR spectrum of $MnO_2@MOF-5$ which may be due to overlapping of Mn-O vibration with secondary building unit of MOF-5, coordinated [Zn₄O]⁶⁺ cluster.

The morphology and composition of MOF-5 and MnO₂@MOF-5 was studied by scanning electron microscopy and SEM based EDX, respectively. The SEM images are shown in figure 3, which shows that MOF-5 has grown in crystalline form having rectangular shape (Figure 3(a-b)). Figure 3(d-e) shows the SEM images of MnO₂@MOF-5 composite which indicates that MnO₂@MOF-5 also have morphologically smooth surface of the crystals. The EDX spectrum of MOF-5 and MnO₂@MOF-5 are shown in figure 3c and 3f, respectively. Figure 3c shows that the EDX spectrum of MOF-5 contains all the three elements (Zn, C and O) of MOF-5. Similarly, figure 3f shows that EDX spectrum of MnO₂@MOF-5 composite contain all the four elements (Zn, Mn, C and O) and indicated successful incorporation of MnO₂ into MOF-5.



Figure 3: (a-b) SEM images, (c) EDX of MOF-5and (d-e) SEM images, (f) EDX of MnO2@MOF-5

The elemental mapping analysis of MOF-5, as shown in figure 4(a-d), shows that there is uniform and homogeneous distribution of elements and it has grown in rectangular shape. The Figure 4a-4c

show elemental mapping of MOF-5 for Zn, C and O respectively and 4d is the mix mapping of metal organic framework, MOF-5, which indicated that all the basic elements are uniformly and homogenously distributed in the sample. Similarly, figure 4e-4g show elemental mapping of MnO₂@MOF-5 for Mn, Zn and O respectively and 4h is the mix mapping of MnO₂@MOF-5, which indicated that nanoparticles of MnO₂ has been successfully incorporated within MOF-5 as it can be observed that Mn is uniformly distributed in the composite.



Figure 4: Elemental mappings of MOF-5and MnO2@MOF-5

The optical properties of synthesized samples were studied by UV-Visible absorption spectrophotometry. As shown in figure S3, MOF-5 has maximum absorption in UV-region at 265 nm due to $\pi \rightarrow \pi^*$ electronic transition of 1,4-BDC linkers and with no significant absorption in visible light. After incorporation of MnO₂ into MOF-5, its λ_{max} shifts from 265 nm to 275 nm. Furthermore, a new absorption peak having λ_{max} around 400 nm has introduced with increase in absorption in the visible region. Such a modification increases the charge separation because photogenerated electron efficiently transfer from organic ligand 1,4-BDC to [Zn₄O]⁶⁺ clusters of MOF-5 and then to MnO₂, similar to UIO-66(NH₂) and Ti-MOF-NH₂. The increase of charge separation has improved the photoelectrochemical activity. The band gap of a semiconductor can be calculated from Tauc plot by linear extrapolation of absorption edge by using following equation

$$\alpha h v \propto (h v - E_g)^{0.5} \tag{4}$$

Where α =absorption coefficient, h= Planck's constant, v= wavenumber

The band gap of MOF-5 is determined to be 3.70 eV, which is in accordance with literature reported previously [38]. While the band gap of $MnO_2@MOF-5$ is 2.47 eV, which indicated that incorporation of MnO_2 nanoparticles brings the band gap of MOF-5 into visible region. Hence, it increases the absorption of visible light and improves the photoelectrochemical activity of

MnO₂@MOF-5. The mechanism of increase of charge separations proposed in schematic representation as shown in Figure 5.



Figure 5: Schematic representation of increased charge separation and water splitting by MnO2@MOF-5

Photoelectrochemical OER analysis

Firstly, photoelectrochemical studies towards OER are determined by cyclic voltammetry. CV curves of these electrodes are recorded using 1.0 M NaOH electrolyte both in presence of dark and under visible light at different scan rates (10, 20, 30, 40, 50 and 100mVs⁻¹) for MOF-5/NF and (5, 10, 20, 30, 40, 50, 60, 70, 80, 90 and 100mVs⁻¹) for MnO₂/NF and MnO₂@MOF-5/NF. It is found that all the working electrodes delivered almost zero current density in dark due to absence of OER activity. Figure 6(a) shows that MOF-5/NF generates no significant current density at various scan rates within RHE potential range of 0.2 to 1.2 V. Figures 6(b) and (c) showed CV curves for MnO₂/NF and MnO₂@MOF-5/NF electrodes under RHE potential range 1.0 to 1.6 V, in dark as well as in visible light. It is investigated from the CV curves of MnO₂/NF and MnO₂@MOF-5/NF that in the presence of visible light prominent increase in current density is observed, due to OER activity. However, pre-OER oxidation-reduction peaks are appeared in CV curves of MnO₂/NF and MnO₂@MOF-5/NF due to redox reaction of Ni(II)/Ni(III) [39]. In MnO₂/NF the oxidation and reduction peaks were appeared in range between 1.44 to 1.50 V and 1.38 to 1.34 V, respectively. The CV curves of MnO₂@MOF-5/NF electrode showed that there are two oxidation peaks at various scan rates (10 to 100mVs⁻¹), first between 1.35 to 1.41V and then between 1.40 to 1.47 V, and only one reduction peak is observed between 1.28 to 1.20 V. It is revealed from the comparison of CV curves of MOF-5/NF, MnO₂/NF and MnO₂@MOF-5/NF that MnO₂@MOF-5/NF has higher current density as compared to pure MOF-5/NF and MnO₂/NF due to hetero-junction formation between MnO₂/NF and central metallic cluster of MOF-5. Thus, synergistic effect and hetero-junction formation played a significant role to enhance the OER activity of MnO₂@MOF-5/NF as compared to all other synthesized samples. Inset of Figure 6 (b and c) represented the anodic and cathodic peak current density with respect to scan rate of MnO₂/NF and MnO₂@MOF-5/NF, respectively.



(c).Inset is Peak current density vs scan rateat different scan rates

The catalytic activity of MOF-5/NF, MnO₂/NF and MnO₂@MOF-5/NF electrodes are further evaluated by linear sweep voltammetry (LSV) in the presence of visible light in 1 M NaOH electrolyte, as shown in figure7a. As in CV, the anodic oxidative peak is observed in LSV of both MnO₂/NF and MnO₂@MOF-5/NF due to oxidation of Ni²⁺ \rightarrow Ni³⁺ from Ni-foam. The LSV curve of MOF-5/NF electrode indicated that there is no significant generation of current density, even at very high overpotential it generated just 0.46 mAcm⁻² current density. For comparison between MnO₂/NF and MnO₂@MOF-5/NF, overpotential required to achieve 2 mAcm⁻²current density is considered. It is observed that MnO₂@MOF-5/NF required low overpotential of 284 mV to achieve 2 mAcm⁻² current density. Whereas MnO₂/NF required 324 mV overpotential for 2 mAcm⁻². It is observed that MnO₂@MOF-5/NF delivered the benchmark of 10 mAcm⁻² at overpotential of 324 mV, which could be comparable and even less then some of reported Mn-based and other 3d transition metal based catalyst for OER, represented by Table 1. Furthermore, to understand the kinetics of MnO₂@MOF-5/NF towards OER, Tafel plot is derived from LSV. The calculated Tafel slope values for MnO₂@MOF-5/NF is just 71 mVdec⁻¹, which is lowered then MnO₂/NF (157 mVdec⁻¹) and previously reported Mn-based materials such as MnO₂/Ni-Co carbonate precursor/NF (95 mVdec⁻¹) and MnO₂/NiCo₂O₄/NF (139 mVdec⁻¹) [40]. The lower Tafel slope value for MnO₂@MOF-5/NF indicated that it has more favorable electron transferred and improved catalytic activity as compared to MnO₂/NF and its rate-determining step for OER was the first electron transferred as shown below,

$$M + OH^{-} \rightarrow MOH + e^{-}$$
 (5)

Where, M is the catalytic active site [37-38].



Figure 7: (a) LSV curves of MOF-5/NF, MnO₂/NF and MnO₂@MOF-5/NF; (b) Tafel plot for MnO₂/NF and MnO₂@MOF-

5/NF

Table 1: The comparison of OER activity of $MnO_2@MOF-5/NF$ with previously reported Mn-based and other 3d transition metal based OER catalyst.

Catalyst	Overpotential (mV) for	Reference
	10mAcm ⁻² current density	
Co ²⁺ /MnO ₂	360	ACS Catal. 2016, 6,7739.
Ni ²⁺ /MnO ₂	400	Angew Chem. Int. Ed. 2016, 55,10381
α- MnO ₂	490	J. Am.Chm. Soc. 2014, 136, 11452.
Metal-ion doped MnO ₂	390	Adv. Funct. Mater. 2017, 27, 1704083
CoMnP	330	J. Am.Chm. Soc. 2016, 138, 4006.
AuNi heterodimers	350	Small 2018, 14, 1703749
Co ₉ S ₈ @NOSC-900	330	Adv. Funct. Mater. 2016, 26, 4712.

Fe-doped mesoporous Co ₃ O ₄	380	Chem. Commun. 2014, 50,10122.
α -MnO ₂ /CDs _{0.25}	352	Electrochim. Acta 2020, 337, 135823
MnO ₂ @MOF-5/NF	324	This work.

From the comparison of LSV curves it can be found that as a result of incorporation of MnO_2 into MOF-5 and due to synergistic effect, $MnO_2@MOF-5/NF$ has improved OER catalytic activity over both pure MOF-5/NF and MnO_2/NF .

Stability of the synthesized materials during OER activity is evaluated by Chronoamperometric studies (Figure 8). It is observed that $MnO_2@MOF-5/NF$ generated constant current density at constant applied voltage of 1.0 V for 6000 seconds in presence of visible light. Stability of MnO_2/NF was also constant during this time but it produced less current density.



Figure 8: Chronoamperometric measurements of MnO₂/NF and MnO₂@MOF-5/NF at 1.0 V applied potential Furthermore, the stability of MnO₂/NF andMnO₂@MOF-5/NF is studied by continuous CV sweeps in 1.0 M NaOH at a scan rate of 100 mV/s for 100 cycles. The CV curves revealed negligible degradation after 100 cycles of CV scanning, which confirmed the stability and durability of these electrodes. Figure S4 (a and b) represented the 1st and 100th cycle of CV curves at 100 mV/s of these electrodes and it revealed almost same curves with negligible difference.

Conclusions

 $MnO_2@MOF-5$ composite has synthesized successfully by in-situ incorporation of pre-synthesized MnO_2 nanoparticles into MOF-5 and used as efficient OER catalyst. The $MnO_2@MOF-5$ composite has shown better OER activity as compared to MnO_2 and MOF-5. $MnO_2@MOF-5$ can

exposes active sites more effectively due to incorporation of MnO₂ nanoparticles and synergistic effect. The incorporation of MnO₂ nanoparticles leads a strong electron interaction between 1, 4-BDC (1,4-Benzenedicarbxylate), [Zn₄O]⁶⁺ clusters and MnO₂ and further optimizes the charge transfer, so charge separation increased and electron-hole pair recombination decreased. Furthermore, binder's free formation of MnO₂@MOF-5/NF enhanced the electrical conductivity. The CV, LSV and Chronoamperometric results show that MnO₂@MOF-5 is an efficient OER catalyst and has more stability. This study will encourage for designing more versatile efficient materials by nanoparticles incorporation into MOFs for water splitting, fuel cells, supercapacitors and batteries.

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Conflict of interest:

Authors declare no conflict of interest.

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