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5	Synergy of slippery surface and pulse flow: an anti-scaling solution
6	for direct contact membrane distillation
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#### Abstract

Recent progress on mitigating scaling on hydrophobic membrane distillation (MD) membrane 21 focuses on the design of superhydrophobic/omniphobic surface and process optimization. 22 However, the rationale for scaling resistance is not yet complete. We attempted in this work to 23 unravel the correlation of scaling resistance based on the synergy of slippery surface (via chem-24 physical engineering) and pulse flow (process engineering). Superhydrophobic micro-pillared 25 polyvinylidene fluoride (MP-PVDF) and CF<sub>4</sub> plasma modified MP-PVDF (CF<sub>4</sub>-MP-PVDF) were 26 utilized as the model membranes. We proposed rheometry as a simple quantitative measure for the 27 wetting state in a hydrodynamic environment. Results showed that MP-PVDF possessed pinned 28 wetting and prone to scaling (2000 mg/L CaSO<sub>4</sub> solution) in both steady and pulse flow. In contrast, 29 the CF<sub>4</sub>-MP-PVDF showed suspended wetting and excellent scaling resistance (at water recovery 30 31 of 60 %, the CF<sub>4</sub>-MP-PVDF surface was still clean without any crystals) under pulse flow, but not at steady flow. At steady flow, feed over-pressure changes the suspended wetting to pinned wetting 32 33 by pushing the water-gas interface into the pillars, thereby resulting in scaling for CF<sub>4</sub>-MP-PVDF. At pulse flow, rhythmic fluctuation in the water-gas interface for CF<sub>4</sub>-MP-PVDF led to sustained 34 scaling resistance. For the first time, we experimentally demonstrated a scaling resistance in 35 DCMD via engineering surface wetting state and process. We envision that this rationale would 36 37 pave the forward-looking strategy for a robust stable MD process in the near future. Keywords: Superhydrophobic; Slippery; Scaling; Suspended wetting; Pinned wetting 38 39

# 41 **1. Introduction**

42 Membrane distillation (MD) is a thermal desalination process which utilizes microporous hydrophobic membrane [1, 2]. As a desalination process that can utilize low-grade heat to treat 43 highly saline brine, MD has been actively explored as a promising technology for saltwater 44 management and zero liquid discharge (ZLD) [3, 4]. Managing hypersaline brine is a key 45 environmental challenge because reverse osmosis (RO) is not applicable when the brine osmotic 46 pressure exceeds the current allowable working pressure of the RO system [5, 6]. In contrast, the 47 48 performance of MD is basically independent of the brine salinity, theoretically allowing MD to achieve a high brine volume reduction up to saturation and crystallization [7]. The inevitable 49 50 challenge of this application is scaling, as the formation or deposition of mineral crystals on the membrane surface, which ultimately leads to a complete process failure [8, 9]. 51

Scaling poses a detrimental effect on MD performance by blockage of membrane pores and reduction in permeance due to homogeneous and heterogeneous nucleation of inorganics [10]. Homogeneous nucleation is difficult which requires supersaturation and sometimes seeding. Comparably, heterogeneous nucleation is relatively easy to occur at the membrane surface. The strategy to largely constrain the formation of crystal scaling on the membrane surface is to increase the membrane hydrophobicity, thus reduce the surface energy or increase the heterogeneous nucleation energy [11].

Classical nucleation theory assumes that low surface energy is the most important governing
factor for mitigating scaling [12-15]. Recently, superhydrophobic surface and omniphobic surfaces

were researched as an important means for scaling and fouling mitigation [16-20]. Besides efforts 61 on the membrane surface characteristics, there were also researches focused on process controls, 62 such as ultrasonic enhancement [21], gas bubbling [22] and microbubble aeration [23]. Results 63 from Lin's group show that a synergistic combination of superhydrophobic membrane and gas 64 purging leads to effective mitigation of membrane scaling [24]. Process Engineering of periodic 65 air backwashing from the distillate side to the feed side has proved to be effective and practically 66 feasible to reduce the membrane scaling [24-26]. However, the process has to be paused and 67 modules have to be drained. An extra blower is required which complicates the system. In this 68 work, we will investigate a facile approach to achieve sustained, stable MD operation with 69 excellent scaling resistance. 70

The rationale of our aim is based on the findings of a hydrodynamic factor of slippage at the 71 72 superhydrophobic membrane/air/liquid interface. The finding was based on precisely designed micro-pillared PVDF membranes (MP-PVDF). We also discovered that a high water contact angle 73 (> 150°) alone did not entail resistance to scaling [27, 28], but slippage at the interface is more 74 75 critical in enhancing the anti-scaling performance. The slippage at the interface is essentially a disruption to the static water layer in the laminar flow scenario [29, 30]. The disruption of the static 76 layer can be achieved by the change of hydrodynamics conditions, which may be achieved in this 77 78 context by the combination of a slippery membrane and a controlled pulse flow.

79 This work exploits the effects of pulse flow in maintaining the direct contact membrane 80 distillation performance subject to gypsum scaling. Tailored made micro-pillared PVDF 81 membranes with and without CF<sub>4</sub> surface modification were compared in the DCMD 82 concentrating process. Membrane surface slip characters, fouling, and flux patterns were examined 83 in order to elucidate the key factors in determining membrane fouling resistance, and thereby to 84 provide some guidance on both membrane and process design.

85

#### 86 2. Materials and methods

## 87 **2.1 Materials and chemicals**

Polyvinylidene fluoride (PVDF, Solvay, Solef 1015) was used for preparing MP-PVDF. The 88 silicon wafer mold with a pillar array was kindly provided by Suzhou Crystal Silicon Electronic 89 and Technology. The pillars show dimensions of 5 um in diameter (D), 10 um in height (H) and 90 10 µm in period (P) [27, 28]. Polydimethylsiloxane (PDMS) and the curing agent (SYLGARD 91 92 184) were purchased from Dow Corning. Diethylene glycol (DEG, AR) and N, N-Dimethylacetamide (DMAc, AR) were supplied by Sinopharm. Calcium chloride anhydrous 93 (CaCl<sub>2</sub>, AR) and sodium sulfate (Na<sub>2</sub>SO<sub>4</sub>, AR) were supplied by Sigma-Aldrich. Procion Blue H-94 5R was purchased from Alfa Aesar. All chemicals were used without further purification. 95

96 **2.2 Membrane preparation** 

97 A tailored-made PDMS mold was replicated with complementary structure from a silicon 98 wafer with a micro-pillar structure following published procedures. It was then used for the 99 preparation of micro-pillared PVDF membrane (MP-PVDF) via a non-solvent induced phase 100 separation (NIPS) method [28].

101 CF<sub>4</sub> plasma treatment was employed to enhance the hydrophobicity of the MP-PVDF using 102 the plasma treatment system(IoN40, PVA Tepla Co. Ltd) based on our previous methods [31-33]. 103 Briefly, the membrane subjected to a pretreatment under argon plasma at 45W/15 s and a CF<sub>4</sub> 104 plasma at a glow discharge of 200W/15 min. The final membrane was denoted as CF<sub>4</sub>-MP-PVDF.

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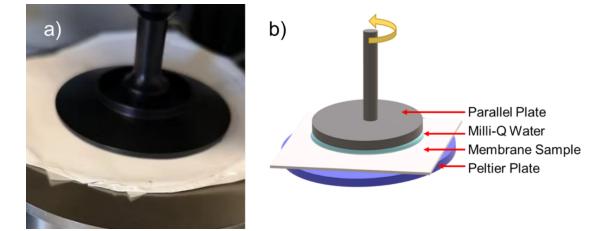
#### 2.3 Membrane characterization

106 The surface wettability of the membranes was determined using a contact angle goniometer 107 (Maist Drop Meter A-100P). The membrane surface morphology was characterized via scanning 108 electron microscopy (SEM) (HITACH TM-1000). The pore size was measured using a capillary 109 flow porometry (Porolux 1000).

# 110 **2.3 Viscometry for wetting state**

The core of this experiment was to accurately measure the apparent viscosity of the water on 111 MP-PVDF and CF<sub>4</sub>-MP-PVDF by a rheometer (AR-2000ex was shown in Supplementary 112 Information S2). As shown in Fig. 1, a membrane was carefully fixed to a Peltier Plate (a 113 114 semiconductor cooling sheet generally considered as a non-slip boundary) using double-sided tape. 1.7 mL Milli-O water was dropped on the center of the sample. The gap between the stainless steel 115 parallel plate (diameter = 60 mm) and membrane surface was fixed at  $h = 600 \mu m$  (auto set by 116 machine) [34]. The experiment was carried out at  $23.0 \pm 0.1$ °C and relative humidity of 64%. 117 118 Temperature and relative humidity are both very important for this measurement due to temperature-sensitive viscosity and evaporation of water. At different shear rates, the torque 119 applied was tracked and the apparent viscosity of the liquid was then determined. The mean value 120

121 of four measurements was calculated as an average over the shear rate of  $150 \text{ s}^{-1}$ .



122

Fig. 1. (a) Photo of the test for apparent viscosity; The white was the membrane sample taped to a stainless steel plate (thermostatically controlled); black plate above the membrane was the plate connecting to the sensors; with shear rate in the range of 0 to 150 s<sup>-1</sup>; (b) Schematic of the components for measurement. Milli-Q water was used as the probe liquid. The excess amount of liquid was carefully removed using a tissue.

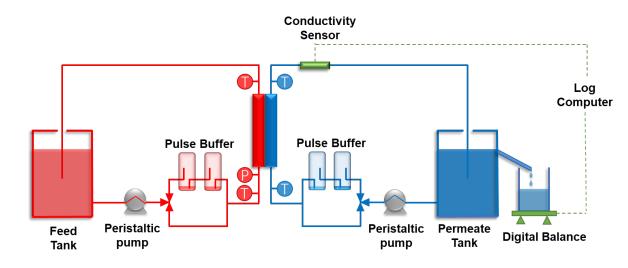
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#### 129 **2.4 Direct contact membrane distillation**

A bench-scale direct contact membrane distillation (DCMD) unit (Fig. 2) was used to evaluate the scaling behavior with and without the pulse by peristaltic pumps and buffers. The peristaltic pump rotates at 56 rpm in the feed side, which means the frequency is 2.8 Hz (56 r/min and three pulses per rotation). By applying two buffers, the pulse from the peristaltic pumps was suppressed (Fig. 2). The pulse buffer was an empty container of about 250 ml. After introducing liquid, the buffer container was partly filled with liquid (liquid volume around 1/10 of the total). Due to the large air volume, the pulse was suppressed and the flow after buffer became then steady. By trialand-error we figured out that two buffers were sufficient to create a rather steady flow. The feed was a calcium sulfate solution (2000 mg/L, prepared by mixing 14.7 mM CaCl<sub>2</sub> and 14.7 mM Na<sub>2</sub>SO<sub>4</sub> solution). The original Saturation Index (SI) of the solution was 0.09. The SI was defined as the logarithm of the ratio of the Ion Activity Products (IAP) divided by the equilibrium constant (K<sub>sp</sub>), respectively [35, 36]:

142 
$$SI = \log\left(\frac{IAP}{K_{sp}}\right)$$

The apparent SI with the different water recovery was shown in Figure 3 and calculated by using 143 OLI Stream analyzer (OLI Systems, Inc., Morris Plans, NJ). The effective area of membranes for 144 evaporation was 1000 mm<sup>2</sup> (50 mm in length, 20 mm in width and 3 mm in channel height). The 145 same flow velocity of feed and permeate was maintained at 0.17 m/s. The resulting solution was 146 supersaturated at 70 °C of 1.2 L. The In this experiment, the surface with micro-pillars was in 147 148 contact with the feed. The temperature of feed and permeate inlet solutions was maintained at 70  $\pm 0.1$  °C and 20  $\pm 0.6$  °C, respectively. The water vapor flux (J, kg/m<sup>2</sup>·h) across the membrane was 149 examined by measuring the permeate weight over time. 150



151

Fig. 2. Schematic diagram of the DCMD test unit. Membrane surface with pillars was facing the
feed solution. Both the feed and permeate flow rate was 600 mL/min. The rotating speeds of the
feed and permeate peristaltic pumps were measured to be 56 and 58 r/min, respectively.

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#### 156 **3. Results and discussion**

157 **3.1 Membrane characteristics** 

Table 1 lists the main surface properties of MP-PVDF and CF<sub>4</sub>-MP-PVDF. CF<sub>4</sub>-MP-PVDF has a mean flow pore size of  $0.201 \pm 0.013 \mu m$ , higher than the pristine MP-PVDF ( $0.120 \pm 0.005 \mu m$ ), mainly caused by the plasma etching. Due to the hydrophobic nature of PVDF and the pillar morphology, a very high static contact angle of  $155.5 \pm 1.3^{\circ}$  was observed for the MP-PVDF. Besides the CF<sub>4</sub>-MP-PVDF showed an even higher static water contact angle of  $165.5 \pm 1.7^{\circ}$ , which is ascribed to the CF<sub>4</sub> plasma treatment that has reduced the surface energy.

Table 1 shows that the sliding angle of MP-PVDF ( $15 \pm 2.3^{\circ}$ ) is higher than that of CF<sub>4</sub>-MP-164 165 PVDF (3.0  $\pm$  0.2°). Due to the etching effect and the deposition of the CF<sub>4</sub> plasma, the surface energy of CF<sub>4</sub>-MP-PVDF was lower than MP-PVDF [28, 31]. Moreover, lift-up effect reduces the 166 contact area between water and CF<sub>4</sub>-MP-PVDF (or water-membrane-air three phase contact lines) 167 [32], thereby lowering the friction force and resulting in a low sliding angle. The low sliding angle 168 is an indication of low friction resistance to the rolling of water droplets. The static contact angle 169 may only reflect the balance of force in the gravity direction. Sliding measurement is very different 170 from the static contact angle in that it reflects the interaction between water molecules and 171

172 membrane surface in shear. The importance of the difference in the sliding angle may be correlated

- 173 to the scaling behavior.
- 174

# 175 Table 1. The properties of MP-PVDF and CF<sub>4</sub>-MP-PVDF surface

	MP-PVDF	CF <sub>4</sub> -MP-PVDF		
Contact angle/°	$155.5 \pm 1.3^{\circ}$	165.5 ± 1.7°		
Sliding angle/°	$15 \pm 2.3^{\circ}$	$3.0\pm0.2^{\circ}$		
Mean flow pore size/µm	$0.120\pm0.005$	$0.201 \pm 0.013$		
Top view	<ul> <li>Ο</li> <li>Ο</li></ul>			

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## 177 **3.2 Membrane scaling and pulse flow**

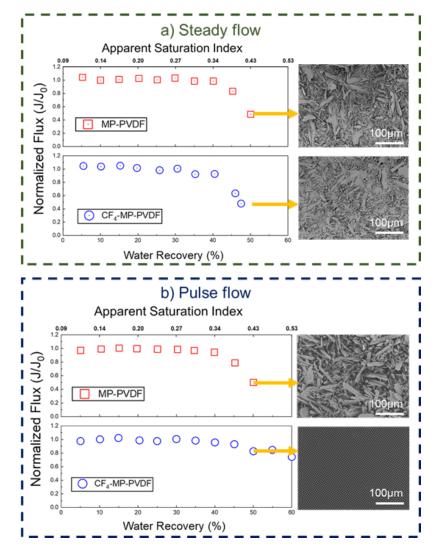
To evaluate the DCMD scaling performance of the two membranes, a supersaturated CaSO<sub>4</sub> solution was used as the feed. For each membrane, steady flow and pulse flow were compared. Fig. 3 shows the DCMD normalized water flux as a function of the water recovery ratio and the eventual membrane SEM images after DCMD experiments termination. As the water recovery increased, the feed solution was cumulatively concentrated followed by supersaturation. Under steady flow, both MP-PVDF and CF<sub>4</sub>-MP-PVDF showed nearly the same induction time of about 13 h until a water recovery of 40%; after this point, MD flux quickly declined due to scaling.
Surface images confirmed that both membrane surfaces were covered by typical CaSO<sub>4</sub>·2H<sub>2</sub>O
gypsum crystals (Fig. 3 a).

The intriguing result was significantly different MD performance of both membranes in case 187 of pulse flow from the peristaltic feed pump. MP-PVDF showed a very similar pattern in the flux 188 verse water recovery curve as that of the steady flow, but the CF<sub>4</sub>-MP-MVDF membrane 189 demonstrated an exceptionally stable MD flux till a water recovery of 60% at the end of the 190 experiment. It should be noted that the water recovery of 60% is the highest achievable result due 191 to limitation of the feed volume; above 60%, air would be pumped together with water to create 192 extra turbulence. At this recovery ratio, the feed salt concentration reached approximately 5000 193 mg/L (2000/(1-60%) = 5000 mg/L), far above the saturation point [37]. 194

It is noteworthy that the flux declined gradually without a significant turning point in contrast 195 to those patterns of membranes that have crystals deposition when the water recovery reached 50%. 196 SEM imaging revealed a clean, nearly intact surface similar to the pristine one with no trace of 197 crystals (Fig. 3 b). We suspected that homogeneous nucleation or heterogeneous nucleation 198 occurred at around 40% water recovery by comparing the flux pattern turning points. For MP-199 PVDF, under steady and pulse flow conditions, the crystals could precipitate or adhere to the 200 201 membrane surface, also for the CF<sub>4</sub>-MP-PVDF under steady flow. But at the pulse flow, for the CF<sub>4</sub>-MP-PVDF, the crystals failed to adhere to the membrane surface though nucleation might 202 have occurred (i.e., the water recovery reached 50%), thus the membrane showed robust scaling 203

resistance under this circumstance. In addition, the permeate conductivity remained unchanged indicating that no wetting occurred even though some of the membranes were scaled intensively. At a water recovery rate of 60%, scaling was found for CF4-MP-PVDF as shown in Supplementary Information Fig. S4. Thus, the slight but gradual decline in MD flux was most probably a result of partial scaling formation. This information was a further support for the scaling resistance of slippery surface with pulse flow.

Based on both thermodynamic and hydrodynamic factors, one would expect that CF<sub>4</sub>-MP-210 PVDF had better scaling resistance than the MP-PVDF: (1) According to the conventional 211 thermodynamic model, very high contact angle and low surface energy corresponds to high scaling 212 resistance [13, 38]; (2) CF<sub>4</sub>-MP-PVDF surface is slippery, thus less prone to scaling due to the 213 non-static liquid surface than MP-PVDF, a non-slippery membrane. However, Fig. 3 indicates that, 214 215 at a static non-pulse flow condition, slippery surface behaved similarly to the non-slippery surface in scaling resistance. Furthermore, it seems that a combination of pulse flow and the slippery 216 surface is sufficient to resist scaling in DCMD. To unravel the above deductions, we need to first 217 quantify the interaction of the water and the membrane surfaces. 218



220

Fig. 3. Normalized water flux of the MP-PVDF (red squares) and CF<sub>4</sub>-MP-PVDF (blue circles) membranes during MD operation at a) steady flow, b) pulse flow. The data for pulse flow was adopted from reference [28]. SEM images of the membrane surfaces were taken when the water recovery reached 50%. Arrows refer to the SEM image taken for the membrane at specific water recovery. Permeate conductivity for all experiments remain constant, thus not shown here. The initial flux J<sub>0</sub> of MP-PVDF and CF<sub>4</sub>-MP-PVDF were 32.0 and 31.1 L m<sup>-2</sup> h<sup>-1</sup> in steady flow and 31.5 and 32.1 L m<sup>-2</sup> h<sup>-1</sup> in pulse flow respectively.

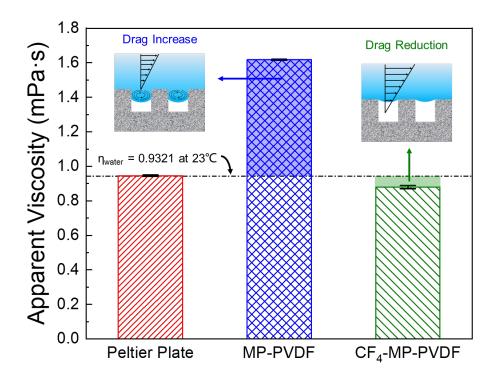
#### 229 **3.3 Wetting state**

Contact angle measurement gives only limited information as the water drops stay static on the 230 membrane surface. Any information on the wetting state at a dynamic condition would not be able 231 to be extracted directly from the results. The sliding angle using a water droplet reflects a 232 hydrodynamic interaction of water and membrane. It reflects the starting of the rolling off of a 233 water droplet; there is still a gap for the interaction of liquid and the membrane at a constant flow. 234 A rheology measurement called viscosimetry was utilized to quantify the hydrodynamic behavior 235 of water at the membrane surface. As shown in Fig. 4, here a simplified interfacial model is 236 introduced by assuming: (1) pillars are not wetted by water; thus, we do neglect the wetting state 237 of the pillars; (2) instead of a water-membrane interface, we identify a water-gas-membrane 238 (polymeric solid area) tri-phase interface (Fig. 4). In rheology measurement, a close-to-reality 239 simulation of the water-gas-membrane tri-phase interface would be constructed using a continuous 240 241 fluidic media. We expect to extract information close to the interface at DCMD.

In Fig. 4, the apparent water viscosity measured at three different surfaces were compared: Peltier Plate, MP-PVDF and CF<sub>4</sub>-MP-PVDF. By definition, the water viscosity is measured on a stainless steel Peltier Plate surface, which is presumably hydrophilic with a static water layer at the water-plate interface during measurement. The measured viscosity of water was  $\eta = 0.946 \pm$ 0.004 mPa·s, within the experimental error for the literature reported value  $\eta_{water} = 0.9321$  mPa·s [39] (the red column in Fig. 4). The relative deviation of the viscosity  $(|\eta_{water}-\eta|/\eta_{water})$  in our work was 1.4%. Choi et al. [40] claimed that the instrumental uncertainty in this measurement was within 0.5%, which required precise control of liquid temperature, the setting resolution of torque and gap size. All these setting controls were very difficult to be realized as we experienced in this work. The other work from Bocquet et al. [41] benchmarked a standard deviation in torque using deionized water of 1.4% at the shear rate of 150 s<sup>-1</sup>. We took this as a more realistic validation of our results.

The second validation is that the experimental deviation of measurement is far below the 254 difference in the viscosity of Peltier Plate, MP-PVDF and CF<sub>4</sub>-MP-PVDF. For the surface, an 255 averaged apparent viscosity of MP-PVDF ( $\eta = 1.617 \pm 0.003$  mPa·s) corresponds to a 71% 256 increase (the light blue area in Fig. 4) to the value measured on the Peltier Plate. Since water is a 257 Newtonian fluid, such a drastic increase in the viscosity means an increase in drag for water by the 258 membrane surface [42]. However, for the CF<sub>4</sub>-MP-PVDF, an averaged apparent viscosity (n 259 =0.879  $\pm$  0.009 mPa·s) (the light green area in Fig. 4) is 7% below the value measured on the 260 261 Peltier Plate. Conversely, this means that the CF<sub>4</sub>-MP-PVDF shows an obvious reduction in the drag for water. 262

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Fig. 4. Apparent viscosities of Milli-Q water measured for a gap of h=600 μm on (i) Peltier Plate
 surface; (ii) MP-PVDF surface; (iii) CF<sub>4</sub>-MP-PVDF surface. The light blue area meant that water
 reserved more drag due to the sagging water-gas interface which induced the vortex between the
 pillars; The light green area meant that water flowed easily on the membrane due to the pinned
 water-gas interface.

 $CF_4$  plasma treatment to the MP-PVDF surface decreases the surface energy due to the fluoropolymer deposition effect [33, 43]. Thus, it is logic that the  $CF_4$  treated membrane showed reduced drag to water in shear. By translating this hydrodynamic factor into a MD operation, the water-gas interface is situated at a suspended state, as shown in Fig. 4. This lift-up state leads to slippage. For MP-PVDF, although the morphology is nearly the same as the  $CF_4$ -MP-PVDF, a

significant increase in the drag in shear represents a severely sagging water-gas interface into the
pillars as shown in Fig. 4. Our observation was also confirmed by the literature reported results
[34]. Much higher drag of the MP-PVDF than that of the Peltier Plate is caused by the hydraulic
vortex flow in liquid below the pillars [44], resulting in significantly high apparent viscosity.

Apparently, this hydrodynamic measure under shear is very similar to the results found in the 281 sliding angle. Table 1 shows that the sliding angle of CF<sub>4</sub>-MP-PVDF is about  $3.0 \pm 0.2^{\circ}$ , lower 282 than that of MP-PVDF ( $15 \pm 2.3^{\circ}$ ). A simple geometry evaluation indicates that the sliding of a 283 water droplet occurs as the friction of the surface to the water droplet is equal to the gravity 284 component parallel to the surface. Thus, we obtain a ratio of friction force of the MP-PVDF to 285 CF<sub>4</sub>-MP-PVDF about 5 (= sin (15°)/sin (3°) = 4.95). Or, a water droplet would find it much easier 286 to roll on a CF<sub>4</sub>-MP-PVDF surface than on a MP-PVDF surface. This phenomenon is strongly 287 related to the wetting state at the surface. 288

As a summary, the wetting state and the hydrodynamic behavior of the water-gas interface are clearly identified in a hydrodynamic condition. MP-PVDF is obviously in a pinned-wetting state. The fluid at the pillar top area remains static, similar to a hydrophilic membrane. CF<sub>4</sub>-MP-PVDF, on the other hand, maintain a suspended wetting state. The water-gas interface remains "floating" on the pillar top, thus contributing to slippage.

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# 295 **3.4 Scaling resistance by synergy of pulse and slippage**

296 The experimental evidence showed that the CF<sub>4</sub>-MP-PVDF is only scale resistant under pulse

flow condition. This result is exceptional. It highlights that membrane surface design and process design are all critical. Very recent research work from Horseman et al [24] showed that air scrubbing would allow stable MD operation, which is in practice of a prevention strategy. The occurrence of nucleation of crystals is not fully prohibited. In this section, we will systematically analyze the scientific origin of scaling resistance from membrane surface design and hydrodynamic control.

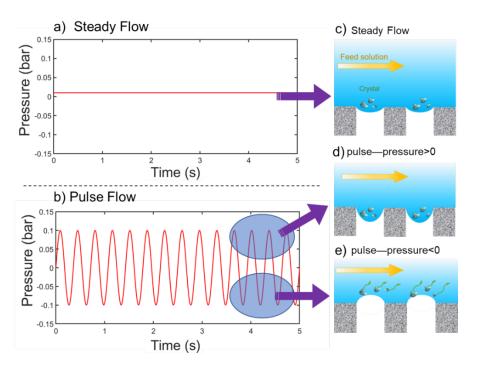
Fig. 3 shows that the MP-PVDF is prone to scale. Combining with the drag measurement, the occurrence of scaling is logic in that pinned wetting corresponds to a non-slip condition, or a static layer at the membrane surface (here means the pillar top). Heterogeneous nucleation would occur as CaSO<sub>4</sub> reaches saturation after a certain induction period of time. However, the CF<sub>4</sub>-MP-PVDF showed nearly the same scaling behavior to the MP-PVDF, although the surface showed reduced drag to water due to suspended wetting and slip (Fig. 4). This groundbreaking evidence means that the slippery surface alone is not sufficient for scaling resistance.

The second result as shown in Fig. 3 is that pinned wetting state (MP-PVDF) plus pulse flow is not scaling resistant either. But, a suspended wetting state with slippage (Fig. 4) appeared to be scaling resistant. This difference might originate from the different gas-water interface.

Fig. 5 outlines the pressure at the upstream of the feed test cell measured using an accurate manometer (Fig. 2). When two pulse buffers were installed, very low positive constant pressure about 0.01 bar was observed (Fig. 5 a); but without buffers, an oscillated pressure between approximately 0.1(over-pressure) to -0.1 bar (under-pressure) was measured (Fig. 5 b). We noticed that the over or under pressures when pulse flow was applied were nearly ten times as that at steady flow. Pressure over ambient at the inlet of the test cell acts as a hydraulic pressure to drive water into the small module inlet and maintain the velocity of the flow inside the small channel between the membrane surface and the cell wall. Because water is an incompressible liquid, this pressure is also applied to the fluid inside the test channel. At the module outlet, the pressure is close to the atmosphere.

This means that the water-gas-membrane interface is not free of external force as always assumed from a thermodynamic model, but rather under low over-pressure and forced to a lower level as schematically depicted in Fig. 5 a. This sagging interface might reduce the slip effect of the CF<sub>4</sub>-MP-PVDF surface. Consequently, the interface becomes similar to that of a MP-PVDF. As a result, the scaling resistance contributed by the suspended wetting and slip is diminished or disappeared, as experimentally demonstrated in the same DCMD performance to the MP-PVDF (Fig. 3).

Fig. 5 d,e schematically described the probable alteration of the water-gas interface in the gaps of pillars in CF<sub>4</sub>-MP-PVDF. Upon over-pressure, the interface sags down into the pillars; at underpressure, the interface lifts up in arching configuration (Fig. 5 e). The oscillation of the interface follows the pulse of the peristaltic pump at a constant period of about 0.36 seconds (= 60/(56\*3)for 56 revolutions per minute produces and three pulses per revolution). This observation corresponds well to previous findings from Duan et al. [45], where the configuration of the watergas interface alters under different pressure. The fluctuation of the water-gas interface on a slippery 337 membrane creates micro-scale hydrodynamic turbulence, which ultimately affects the aggregation 338 of the scalants by homogeneous nucleation, and more importantly gives no residence time for the 339 heterogeneous nucleation on the membrane surface. As a result, a robust sustainable scaling 340 resistance is achieved.



341

Fig. 5. Schematic of scaling behavior for CF<sub>4</sub>-MP-PVDF at steady and pulse flow. a) and b) Pressure
 versus time curve without and with pulse; the water-gas interface c) at steady flow; (d) with the pulse in
 positive pressure and (e) in negative pressure. The continuous line in (b) does not mean exact
 measurement but a linear regression of the pressure changes from negative to positive values.

We attempted to visualize the interface flutuation at over-pressure or under-pressure as shown in Fig. 5. It is unfortunate that we could not take sharp images at the interface yet. Thus, a macroscale image of the surface was taken to support above hypothesis. Fig. 6 shows the snapshots

of the membrane surfaces during MD operation.Obviously, at steady flow, a smooth, "static" membrane surface was observed for both CF<sub>4</sub>-MP-PVDF and MP-PVDF. But, at pulse flow, unique surface bubbles were observed in the CF<sub>4</sub>-MP-PVDF surface only at under-pressure, but not at over pressure.

No visual difference could be identified for MP-PVDF at either under-pressure or over-354 pressure (Fig. 6). The appearing of the bubbles in millimeter size at pulse flow is a result of 355 aggregation of many adjacent small bubbles (Fig. 5 e). After 2 h induction time, no bubbles were 356 seen, but CF<sub>4</sub>-MP-PVDF maintained the scaling resistance for more than 20 h till a water recovery 357 of 50% (Fig. 3). This is an indirect, but sound proof of the "flip-flap" interface hypothesis as shown 358 in Fig. 5. We attempted to using light microscopy to obtain the images of this process. Due to the 359 instrument limit, modification of the experimental setup is under construction and our new results 360 361 would be reported in the near future.

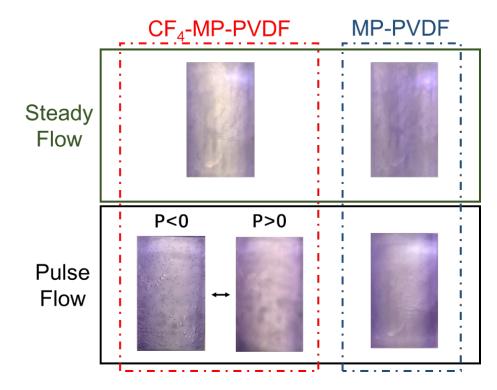




Fig. 6. Observation of the MP-PVDF and CF4-MP-PVDF surface with steady flow and pulse
 flow during MD. Procion Blue H-5R was added to feed at 100 mg/L. At steady flow, no bubbles
 were seen on either surfaces; At pulse flow, only CF4-MP-PVDF surface showed bubbles at
 under pressure.

# 368 4. Conclusions

Synergy of feed flow patterns and wetting state related to slippery effect on the occurrence of scaling in DCMD was investigated based on micro-pillared PVDF membranes. MP-PVDF was found being prone to CaSO<sub>4</sub> scaling at both steady and pulse flows, but CF<sub>4</sub>-MP-PVDF showed scaling at steady flow, but excellent scaling resistance at pulse flow by a clean surface and relatively stable MD flux at a supersaturation. Using rheological measurement, a significant

increase in the apparent viscosity of water for MP-PVDF corresponds to the pinned wetting with 374 a static water layer at the interface, but for CF<sub>4</sub>-MP-PVDF, a slight decrease of apparent viscosity 375 indicates suspended wetting, thus surface slippage. At a steady flow, an over-pressure in feed 376 would forces the water-gas interface to sag into the pillars, thus creating a non-slip condition for 377 both MP-PVDF and CF<sub>4</sub>-MP-PVDF, resulting in severe scaling. At a pulse flow, a pinned wetting 378 in MP-PVDF contributes to a static water-gas interface, but suspended wetting in CF<sub>4</sub>-MP-PVDF 379 leads to a lift-up effect of the water-gas interface and avoids the scaling occurrence. Visualization 380 of the aggregated bubbles at the membrane surface at under-pressure in CF<sub>4</sub>-MP-PVDF further 381 supports the above understanding. The results elaborate a scientific basis for the design of scaling 382 resistant membrane as a combination of superhydrophobic slippery surface plus a pulse flow. Our 383 findings provide a new strategy for the robust scaling MD process in the near future. 384

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# **Supplementary Information**

# Synergy of slippery surface and pulse flow: an anti-scaling solution for direct contact membrane distillation

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#### S1 Wetting stability of pillared surface

We adopted the model from Xue's study [1] on the wetting stability of the submerged superhydrophobic. As shown in Fig. S1 a, we chose the classical Cassie-Baxter (CB) state as a reference, where a liquid-air interface is planar and the air trapped in a pore is at atmospheric pressure  $p_0$  with a volume of  $V_0$  (=  $\pi R^2 H$ ). The pore wall is assumed to be solid, not porous as in our case of membrane. At an external pressure of  $p_l$ , the air-liquid interface protrudes towards the pore and air is pressurizes inside (Fig. S1 b). When the sagging angle  $\theta$  is equal to the advancing angle of the intrinsic material  $\theta^*$ , a critical pressure  $p_i^c$  is obtained. If the pressure is above the critical pressure ( $p_i^c$ ), the liquid-air interface is depinned and advances into the pores (Fig. S1 c).

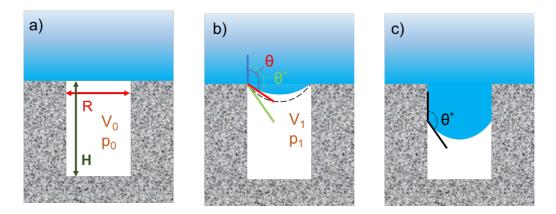


Fig. S1. Schematic illustration of the liquid–air interface in a hole at different state: (a) classical CB state; (b) at external over pressure, (c) after depinned and interface advances into the hole. The dashed line in (b) shows the critical positions of the advancing  $\theta^*$  where the depinning occurs. *R*: pore radius; H: pore depth.

Based on the ideal gas law at a constant temperature, Eq. 1 and the Young-Laplace equation (Eq.

2) relates the meniscus curvature radius and the capillary pressure (the pressure difference across the water-air interface arising from the capillary forces) as

$$V_0 p_0 = V_1 p_1 \tag{1}$$

and

$$p_l = p_1 + \frac{2\cos\theta}{R}\gamma \tag{2}$$

Where *V* and *p* represent the volume and pressure of air in the pore; the subscripts 0 and 1 denote the reference and final states of the air, respectively.  $\gamma$  is the surface tension of the liquid at the constant temperature. R is the radii of the hole. The volume of the liquid protruding into a pore is expressed as,

$$V_{in} = \frac{\pi R^3}{3\sin^3\theta} \left(2 - 3\cos\theta + \cos^3\theta\right) \tag{3}$$

The final volume of the air in the pore is

$$V_{1} = V_{0} - V_{in} = \pi R^{2} H - \frac{\pi R^{3}}{3 \sin^{3} \theta} \left( 2 - 3 \cos \theta + \cos^{3} \theta \right)$$
(4)

Combining Eq. 1-2 and Eq. 4 yields

$$p_{l} = p_{0} / \left( 1 - \frac{R}{3H\sin^{3}\theta} \left( 2 - 3\sin\theta + \sin^{3}\theta \right) \right) + \frac{2\gamma\cos\theta}{R}$$
(5)

Eq. 5 illustrates the equilibrium states as shown in Figure 1b with the three-phase contact line pinned at the top. When the liquid pressure is higher than the initial air pressure  $P_0$ , the water-air interface will protrude downward into the pore. When the sagging angle reaches the advancing angle of the material, the pressure is named as the critical pressure. In other words, submitting  $\theta^*$  for  $\theta$  in Eq. 5, the critical pressure  $p_1^c$  is obtained.

For the pillared surface, corresponding effective geometric and capillary radius is approximate equivalent of the pore pattern surface. Based on Lobaton's work, the effective geometric radius refers to the volume of liquid protruding into each effective pore approximately equal to the volume of liquid between four nearby pillars. For pillars on a square lattice (as shown in Fig. S2), the effective geometric radius is expressed as,

$$R_{eff}^{g} = \left(\sqrt{\frac{\pi}{2f}} - 1\right)R_{p} \tag{6}$$

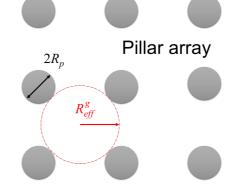


Fig.S2. Schematic illustrating the effective geometric radius  $R_{eff}^{s}$  for a structured surface with a pillar

radius  $R_p$  (top view).

And the effective capillary radius for pillars on a square lattice [2]

$$R_{eff}^{c} = \frac{1 - f}{f} R_{p} \tag{7}$$

Note that the values of  $R_{eff}^{g}$  and  $R_{eff}^{c}$  may differ greatly from each other, depending on the surface structure and solid fraction *f*.

 $R_{eff}^{s}$  and  $R_{eff}^{c}$  were substituted for R and the advancing angle  $\theta^{*}$  is substituted for  $\theta$  in Eq. 5. Then combined Eq. 5-7, Eq. 8 for the pillar substrates can be expressed as

$$p_l^c = p_0 \left( 1 - \frac{f}{\left(1 - f\right)R_p^2 H} \frac{\left(R_{eff}^g\right)^3}{3\cos^3\theta^*} \left(2 - 3\sin\theta^* + \sin^3\theta^*\right) \right) - \frac{2\gamma\cos\theta^*}{R_{eff}^c}$$
(8)

Note that the height of those pillars is H. The parameters to calculate the critical pressures are shown in Table 1. The solid fraction f is equal to a pillar top surface area ratio to a period pattern area. The advancing angle of MP-PVDF and CF<sub>4</sub>-MP-PVDF membranes were taken from the PVDF flat membrane and the CF4 plasma treated PVDF flat membrane. This approximation is taken because of similar surface porous morphology of our tailor-made membrane to the commercial PVDF membrane. The surface tension of water is used.

Membrane	$R_p/\mu m$	<i>H</i> /μm	f	$ heta^*/^{\circ}$	p₀/Pa	γ/ mN·m
						(water at 70°C)
MP-PVDF	2.5	10	0.20	110	101325	64.4
CF <sub>4</sub> -MP-PVDF	2.5	10	0.20	160	101325	64.4

Table S1. The parameters of MP-PVDF and CF<sub>4</sub>-MP-PVDF to calculate the critical pressure

The result of the critical pressure for MP-PVDF is 109.12 kPa and for CF4-MP-PVDF is 131.33 kPa (both absolute pressures). This means that for MP-PVDF, water is depinned because the liquid pressure in membrane module is 0.1 bar (relative pressure) which is higher than the critical pressure; but for CF4-MP-PVDF, water is still suspended because the liquid pressure is lower than the critical pressure.

#### S2 Photo of rheometer AR-2000ex

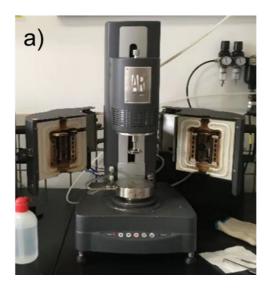


Fig. S3. Photo of the rheometer AR-2000ex used to confirm the wetting state of MP-PVDF and CF<sub>4</sub>-

MP-PVDF

# S3 SEM image of fouled CF<sub>4</sub>-MP-PVDF after MD under pulse flow.

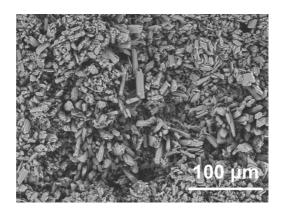


Fig. S4. SEM image of CF<sub>4</sub>-MP-PVDF when the water recovery reached 60% under pulse flow.

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